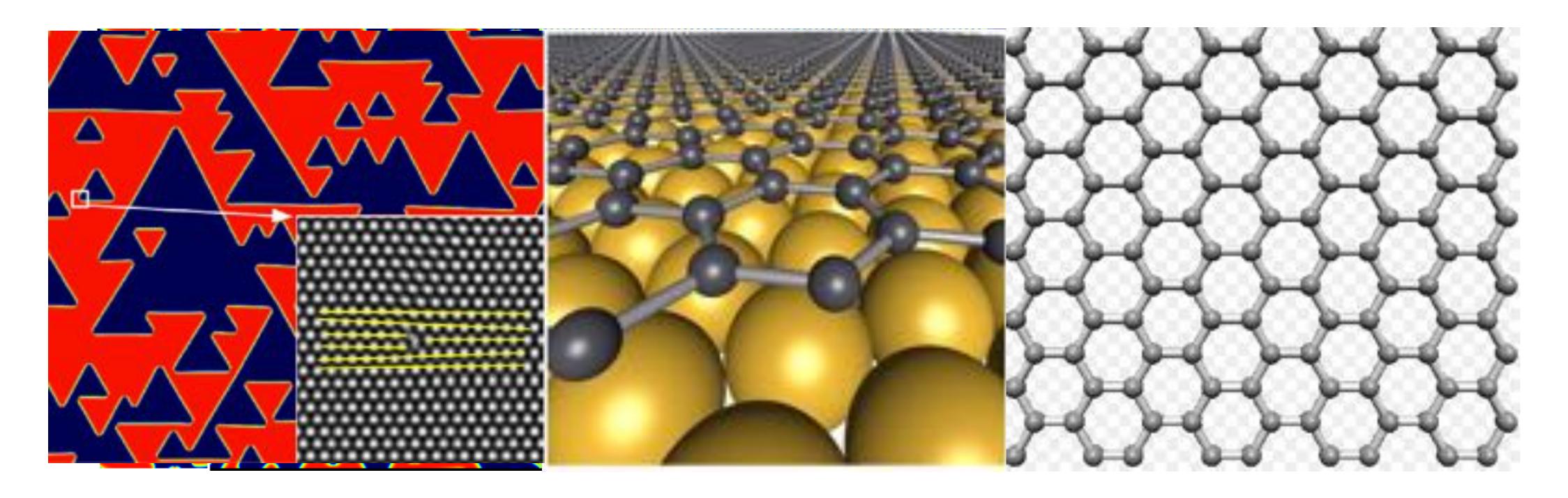




Multi-Scale Modeling of Graphene from Nano to Micron Scales

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[K.R. Elder et al., PRL (2012); PRB (2013); JCP (2016); P. Hirvonen et al., PRB (2016); Sci. Rep. (2017); Z. Fan et al., PRB (2017), Nano Lett. (2017); K. Azizi et al., Carbon (2017)]





Outline of the Talk

- * Introduction to Graphene and Modeling Methods
- * Phase Field Crystal Model
- * Multi-Scale Modeling Strategy for Graphene
- * Large Multigrain Flakes
- * Heat Conduction in Graphene
- * Summary and Conclusions



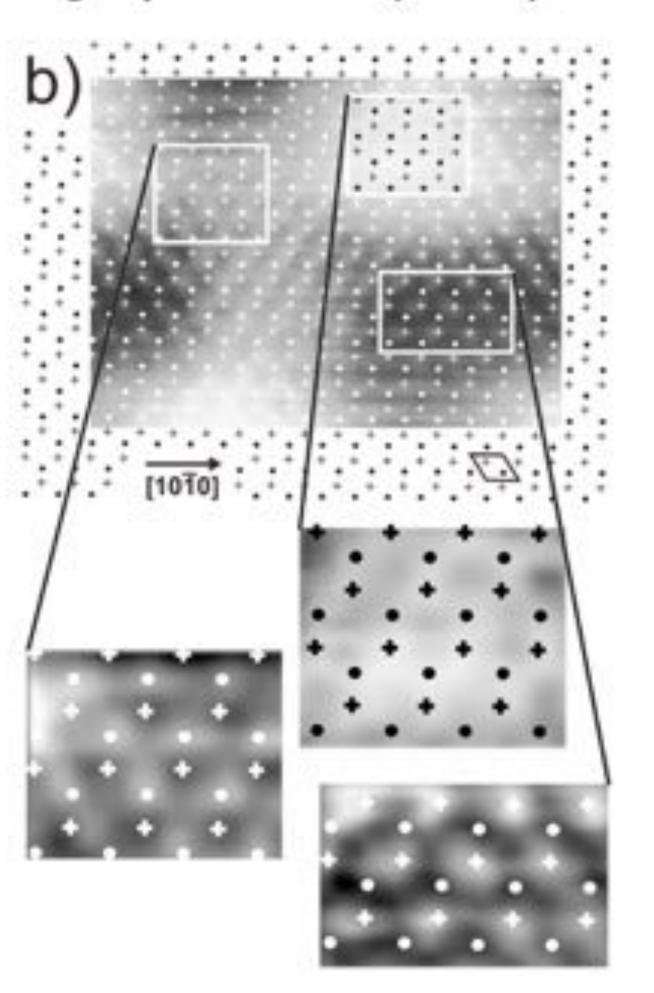


* Graphene is often grown on metal surfaces to achieve epitaxial configurations (usually under tensile stress)

30 Å

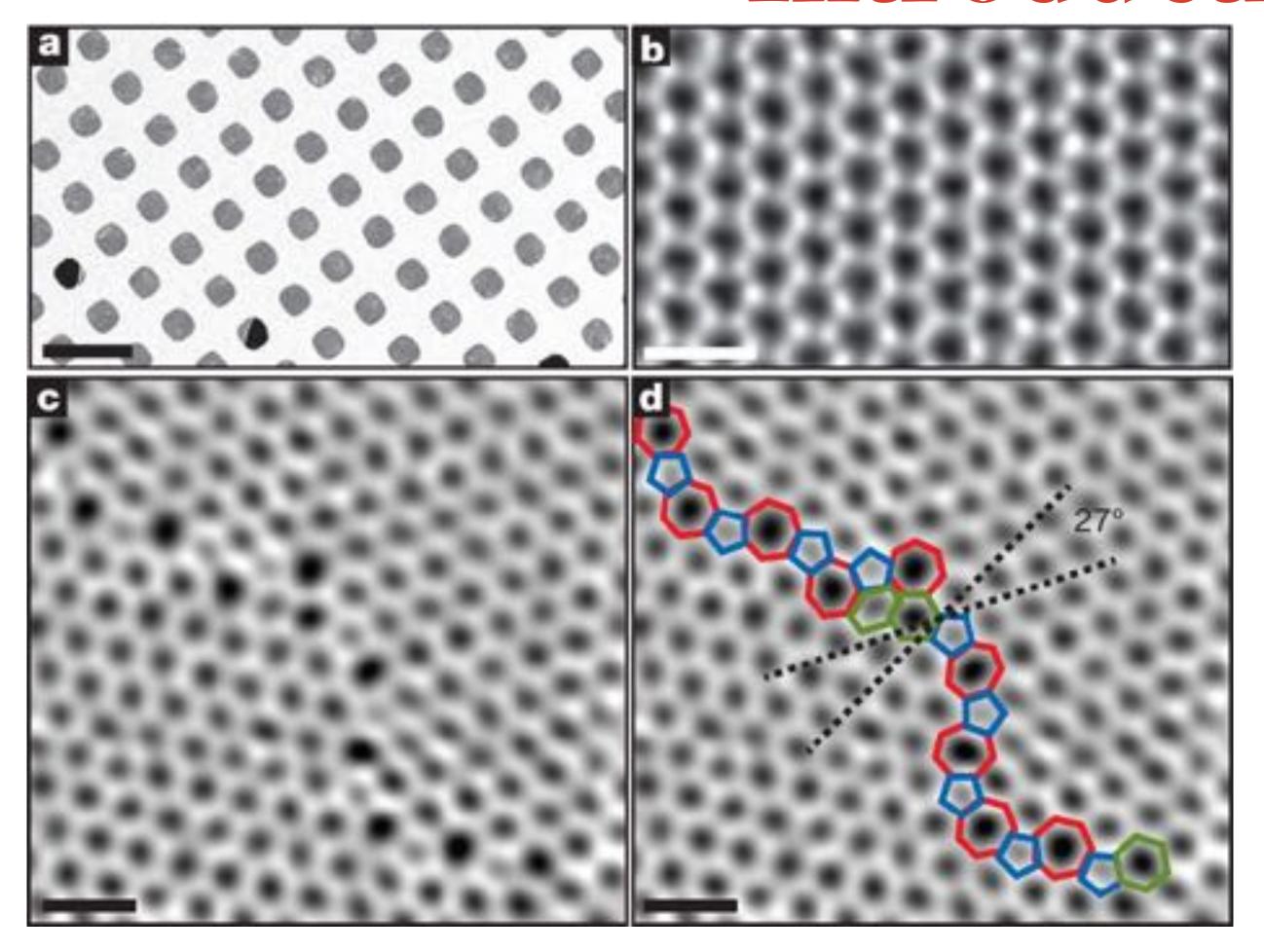
[S. Marchini *et al.*, PRB **76**, 075429 (2006)]

Experiment graphene/Ru(0001)





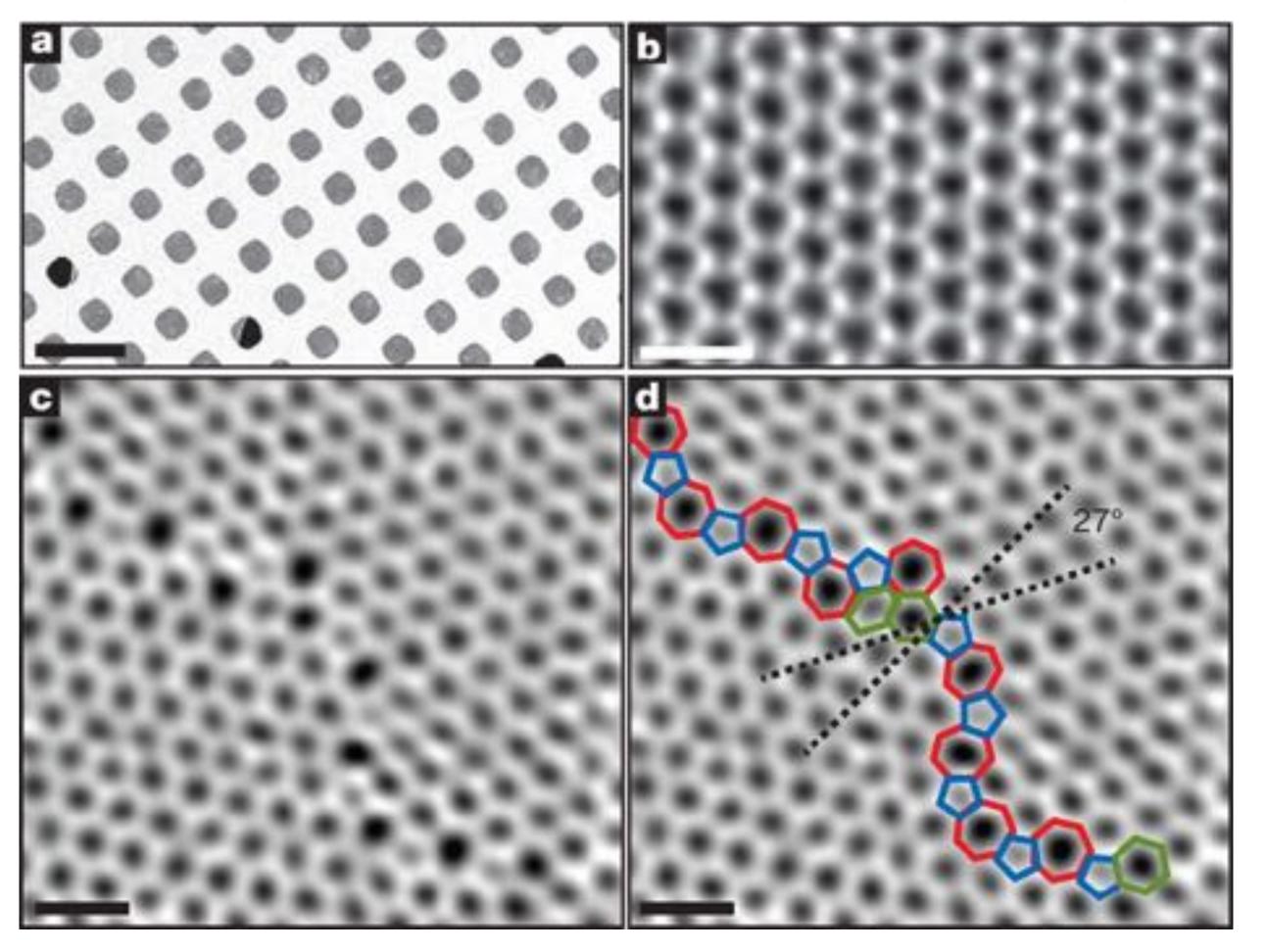




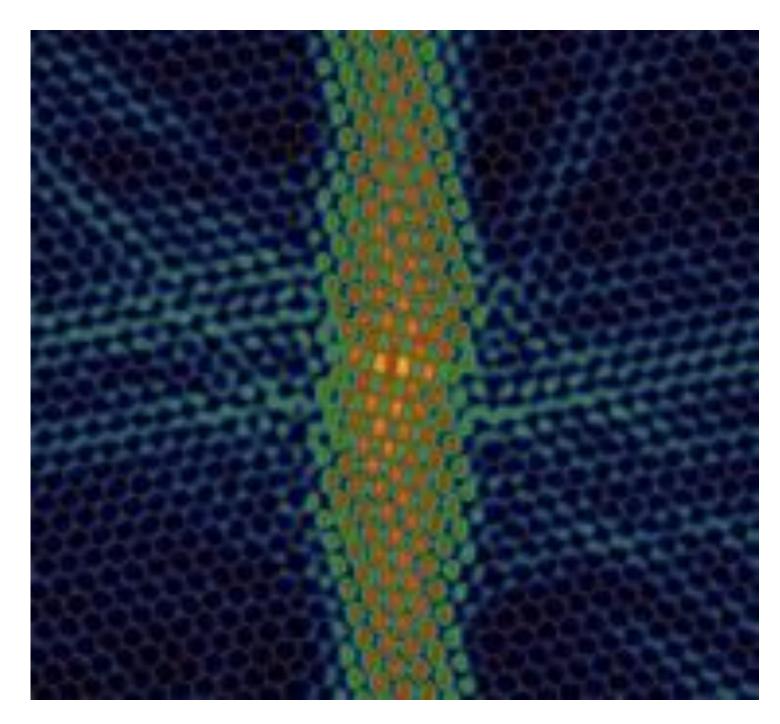
High resolution electron microscopy of graphene flakes [Huang *et al.*, Nature **469**, 389 (2011)]







High resolution electron microscopy of graphene flakes [Huang *et al.*, Nature **469**, 389 (2011)]

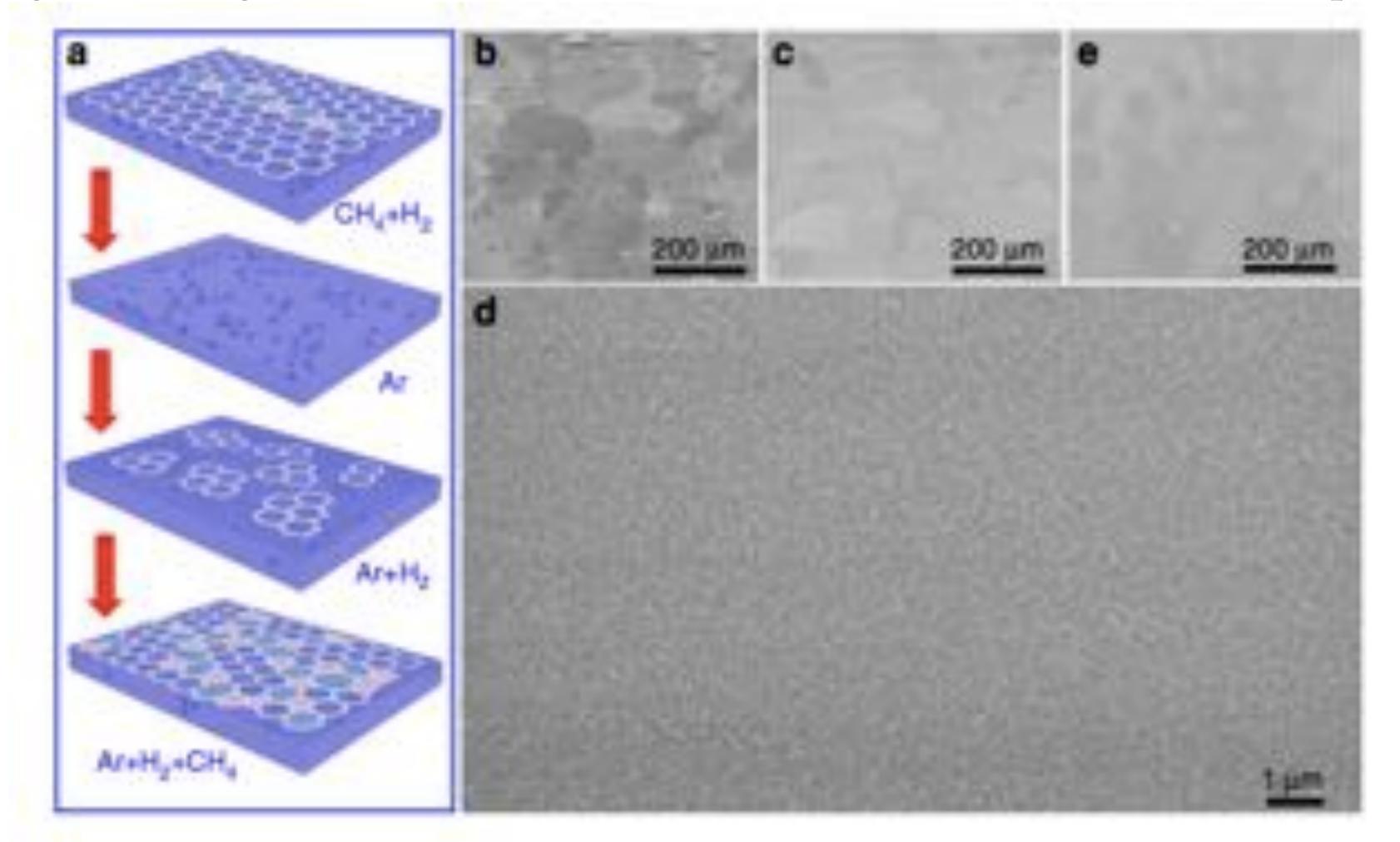


Photonic wave guide between grain boundaries
[Mark *et al.*, J. Nanophoton. **6**, 061718 (2012)]





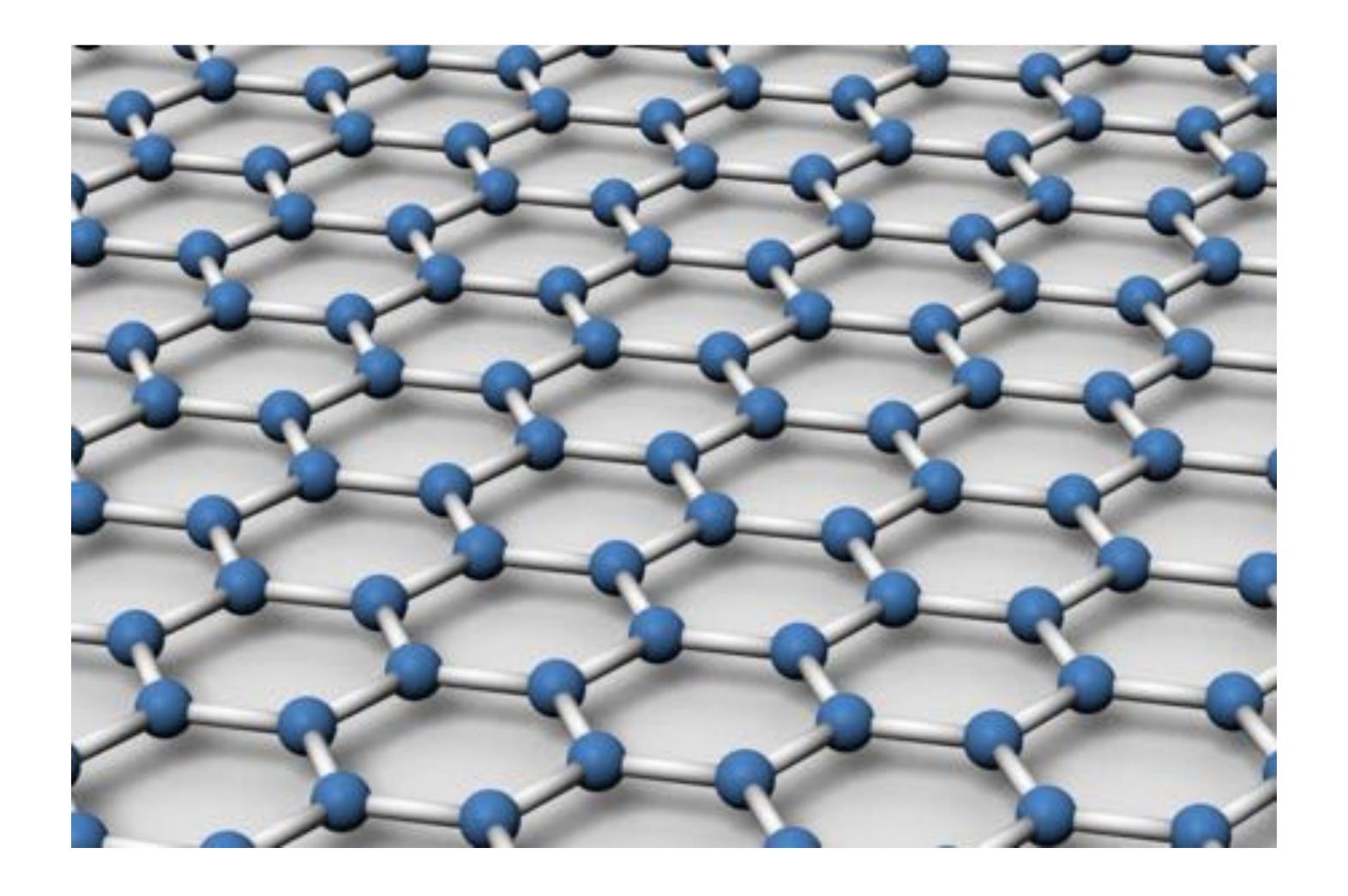
* Controlled growth of grains can now be achieved with modified chemical vapour deposition







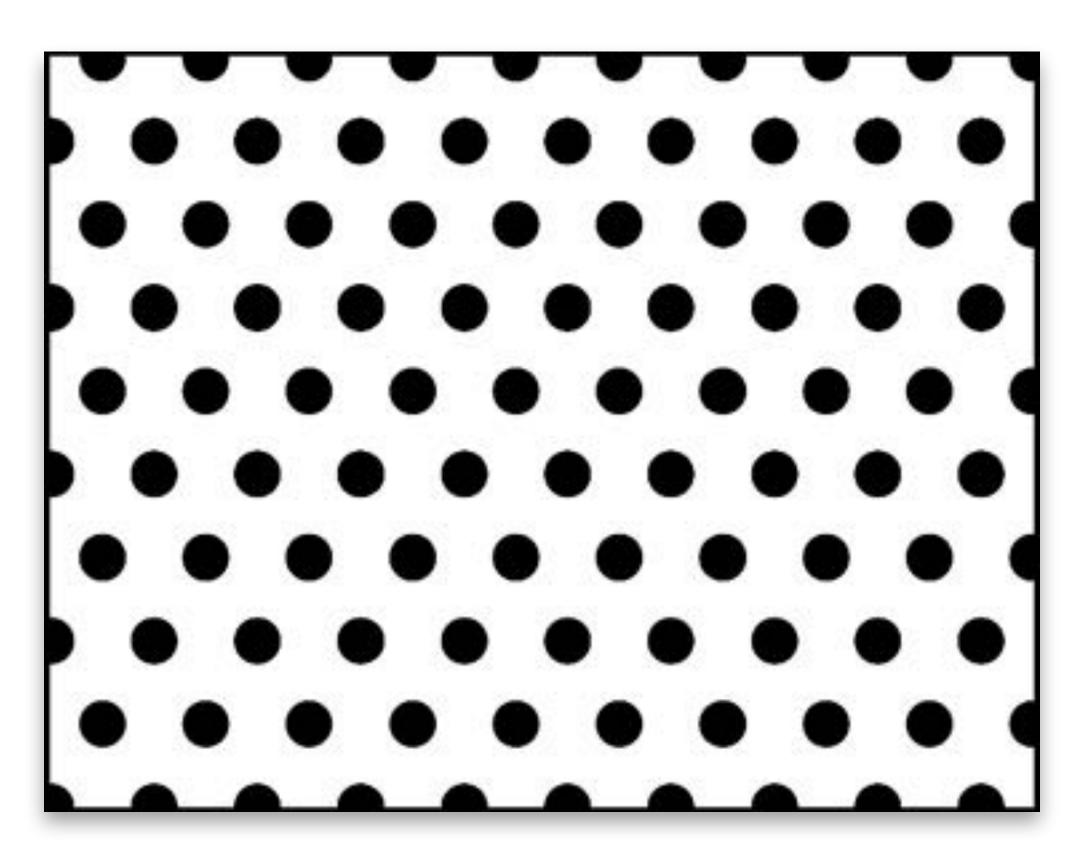
Modeling Graphene

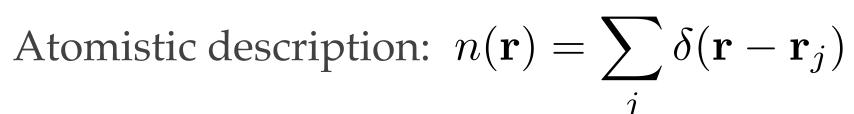


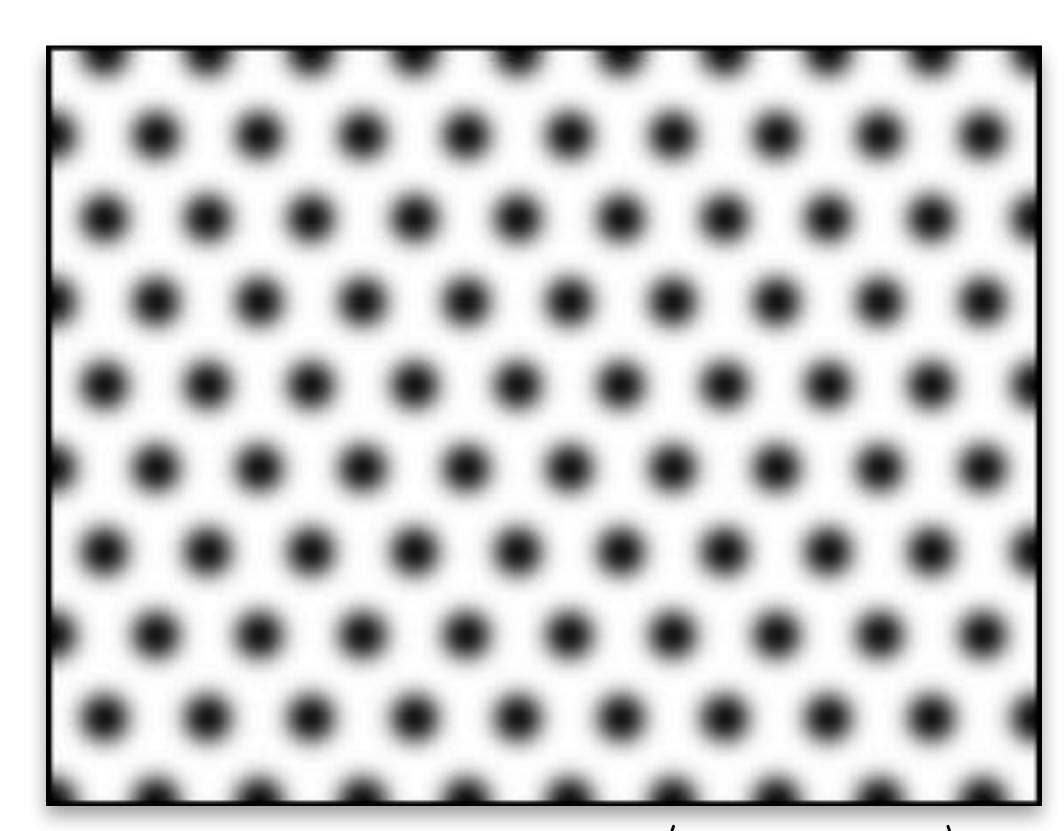




Phase Field Crystal (PFC) Model¹







Statistical description:
$$n(\mathbf{r}) = \left\langle \sum_{j} \delta(\mathbf{r} - \mathbf{r}_{j}) \right\rangle$$

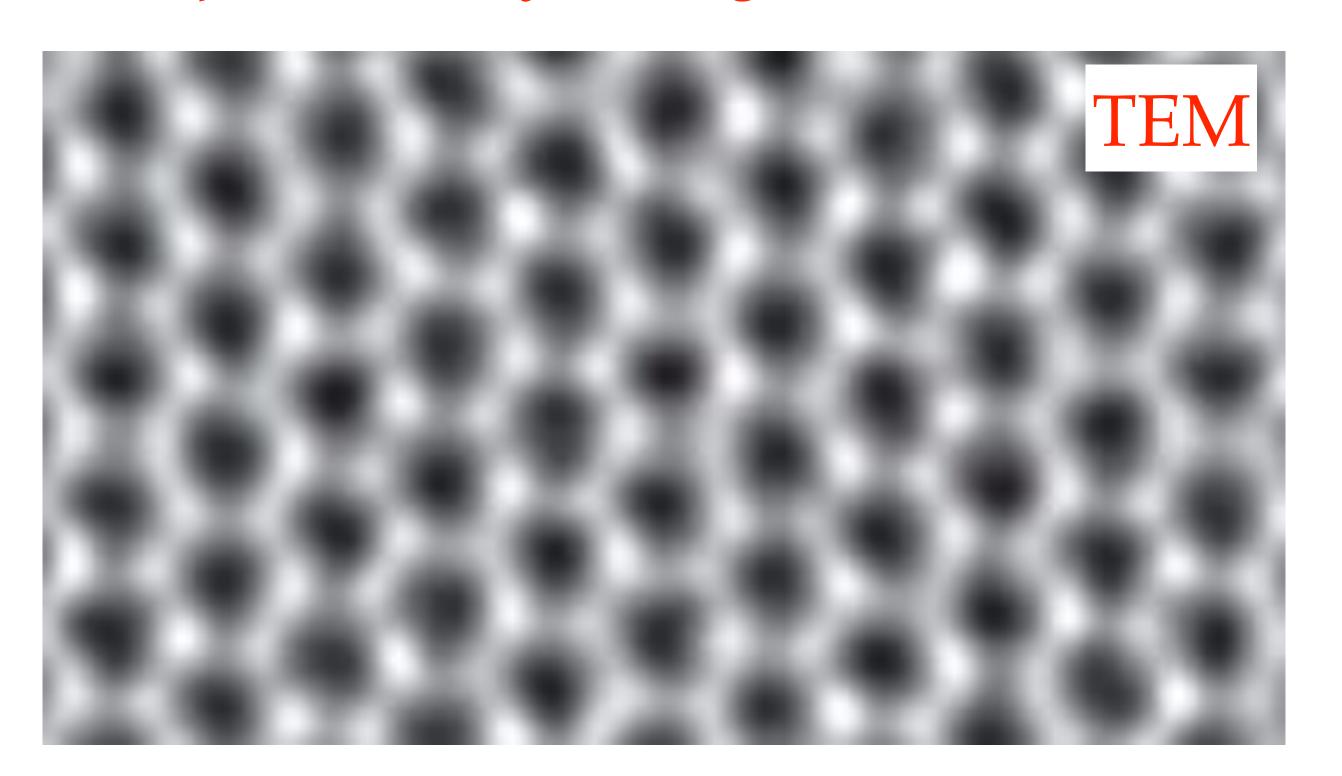


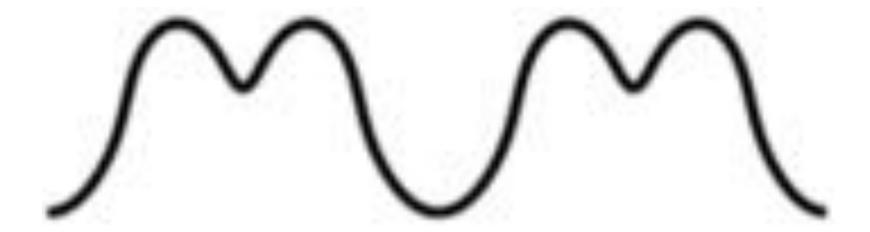


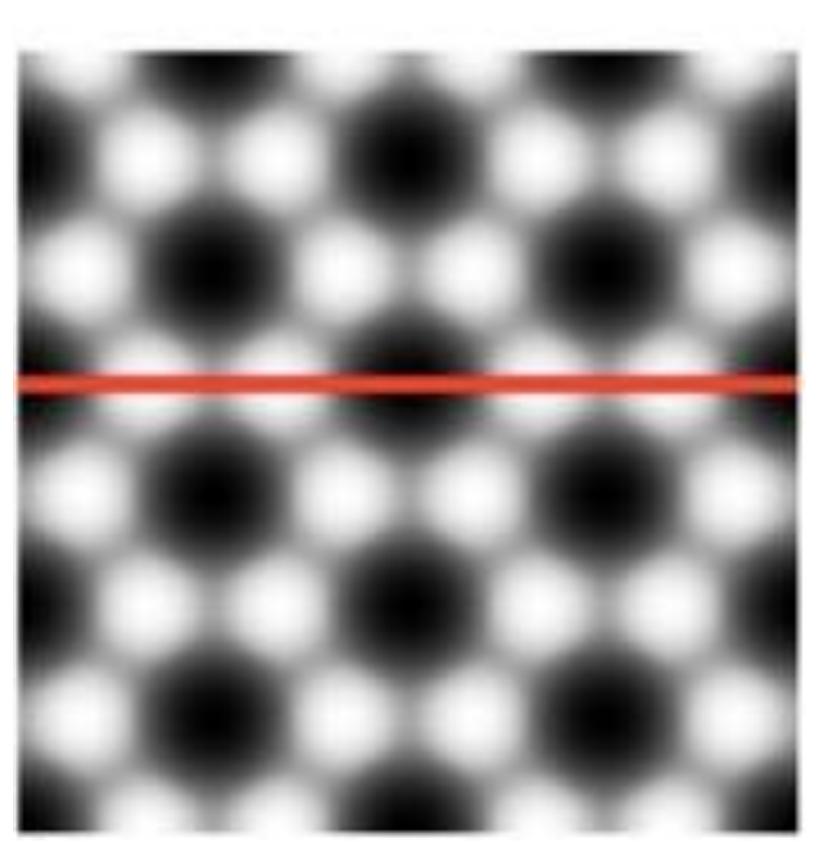
PFC Free Energy

$$F[n(\mathbf{r},t)] = \int d\mathbf{r} \left[\frac{\Delta B}{2} n^2 + \frac{B^x}{2} n(1+\nabla^2)^2 n - \frac{\tau}{3} n^3 + \frac{\nu}{4} n^4 \right]$$





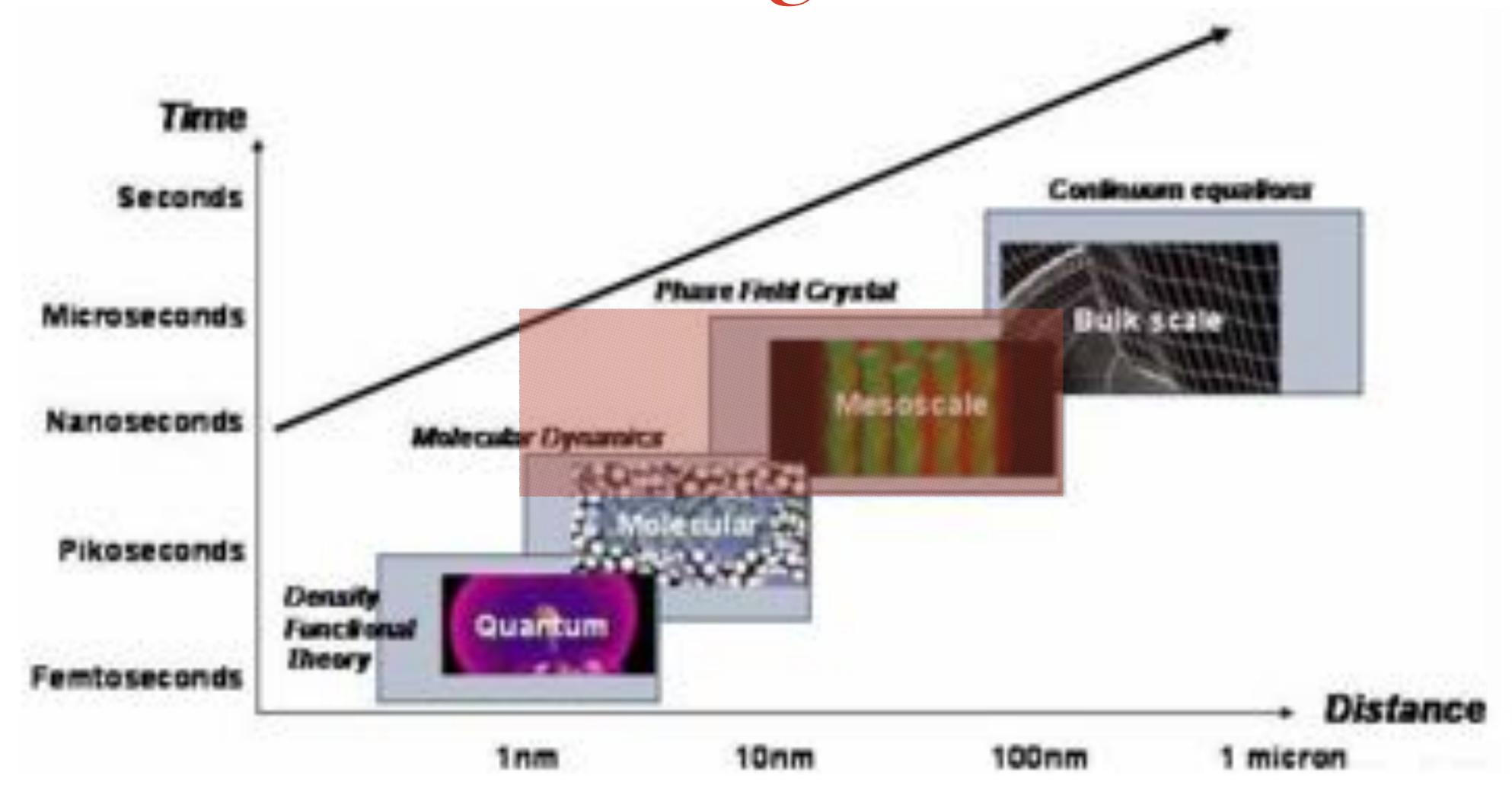








Modeling Scales







PFC Free Energy

$$F[n(\mathbf{r},t)] = \int d\mathbf{r} \left[\frac{\Delta B}{2} n^2 + \frac{B^x}{2} n(1+\nabla^2)^2 n - \frac{\tau}{3} n^3 + \frac{\nu}{4} n^4 \right]$$

- * Describes the energy of the system as functional of the atomistic number density field *liquid and crystalline ground states (and coexistence)*
- * Contains topological defects and elastic excitations
- * Can be derived from classical DFT [Elder & Grant (2004), Jaatinen & T.A-N. (2010)]
- * Usually coupled to dissipative dynamics in time

$$\frac{\partial n}{\partial t} = -\frac{\delta F[n]}{\delta n}$$





Multi-Scale Modeling Strategy for Graphene

- * Use PFC to generate 2D grain boundary configurations
- * Input to all-electron QM-DFT calculations for grain boundary energy and use these to fix energy scale in PFC
- * Generate large-scale, multi-grain PFC samples as input to MD to relax for further calculations (e.g. heat conduction and electric conductivity)





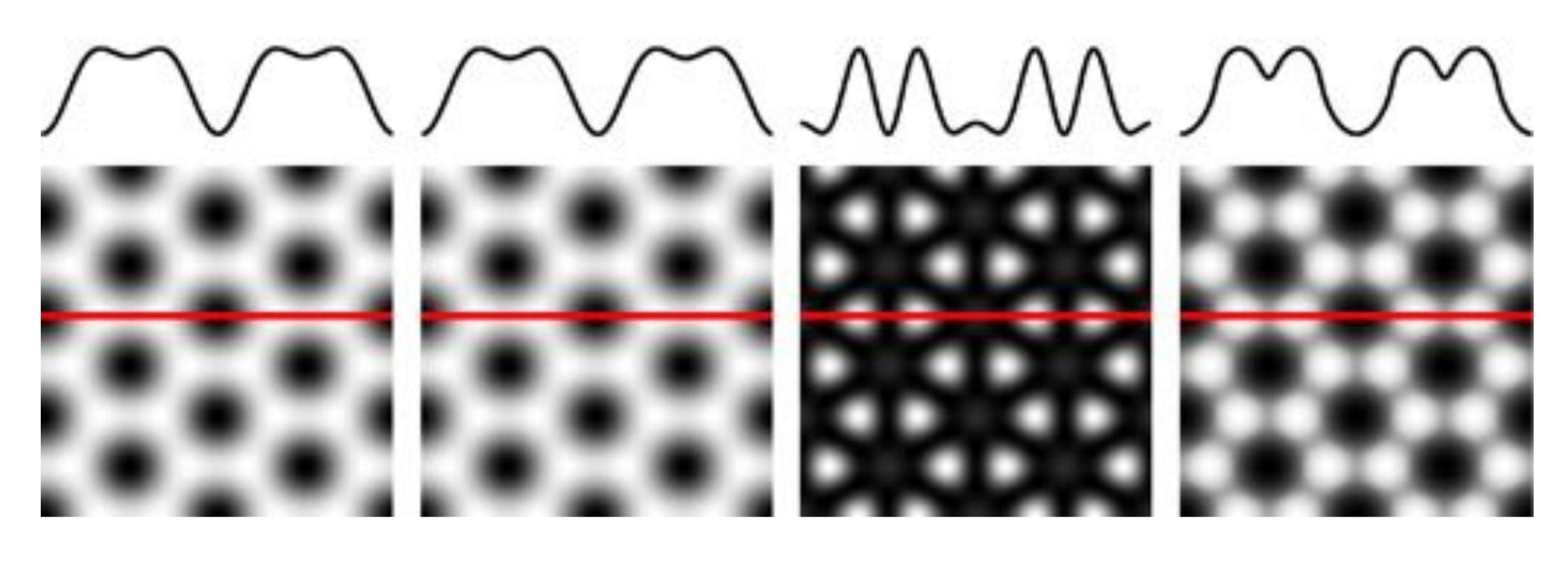
Methodology III: 2D PFC Models

- * PFC1: Standard PFC (412 41 600 atoms)
- * APFC: Amplitude expansion of PFC1 (1 140 5 900 000 atoms)
- * PFC3: Three-mode PFC model (412 41 600 atoms)
- * XPFC: PFC model with two and three-body interactions (412 166 400 atoms) [M. Seymour and N. Provatas, PRB 93, 035447 (2016)]

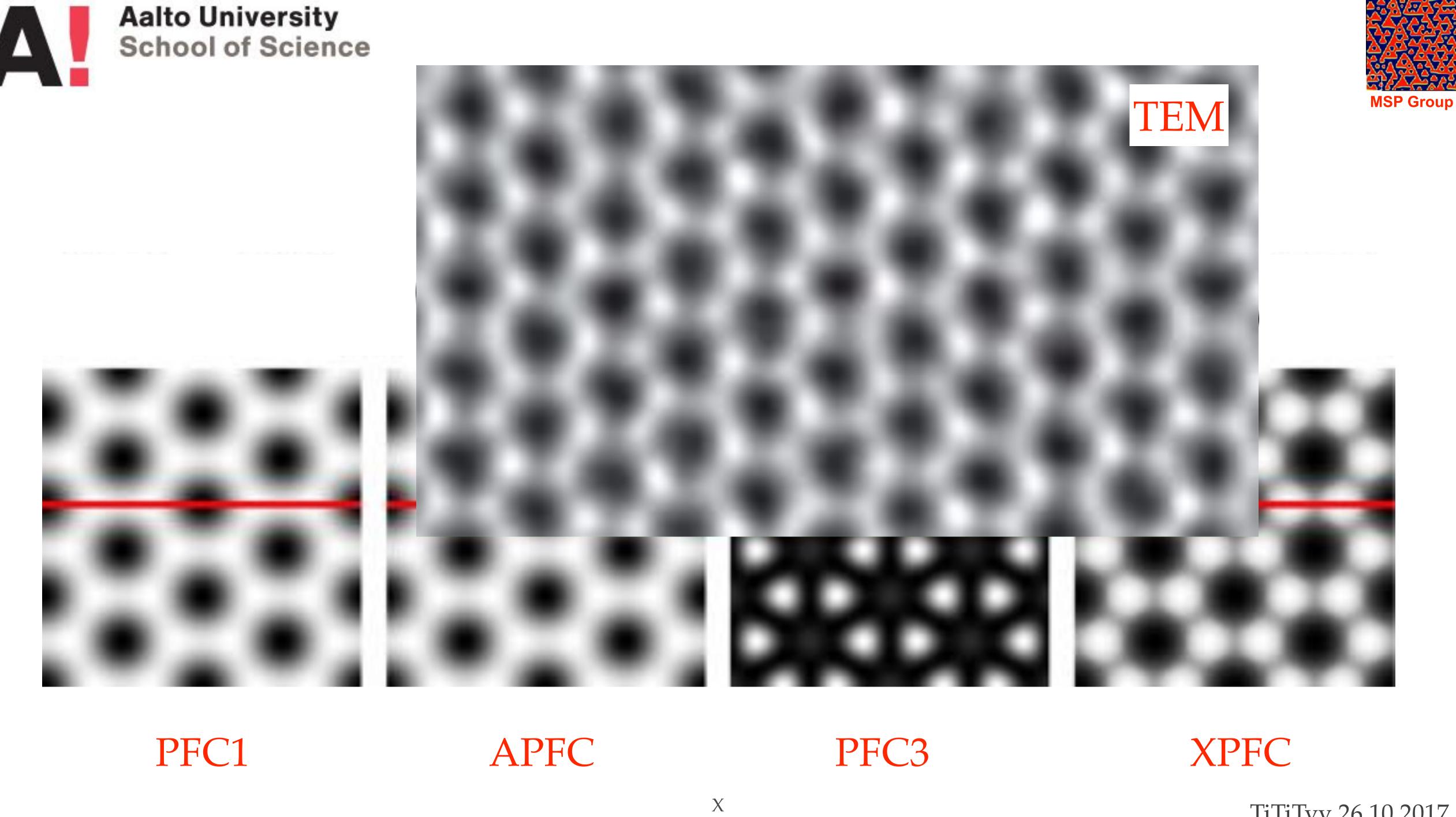




PFC Density Fields



PFC1 APFC PFC3 XPFC



TiTiTyy 26.10.2017







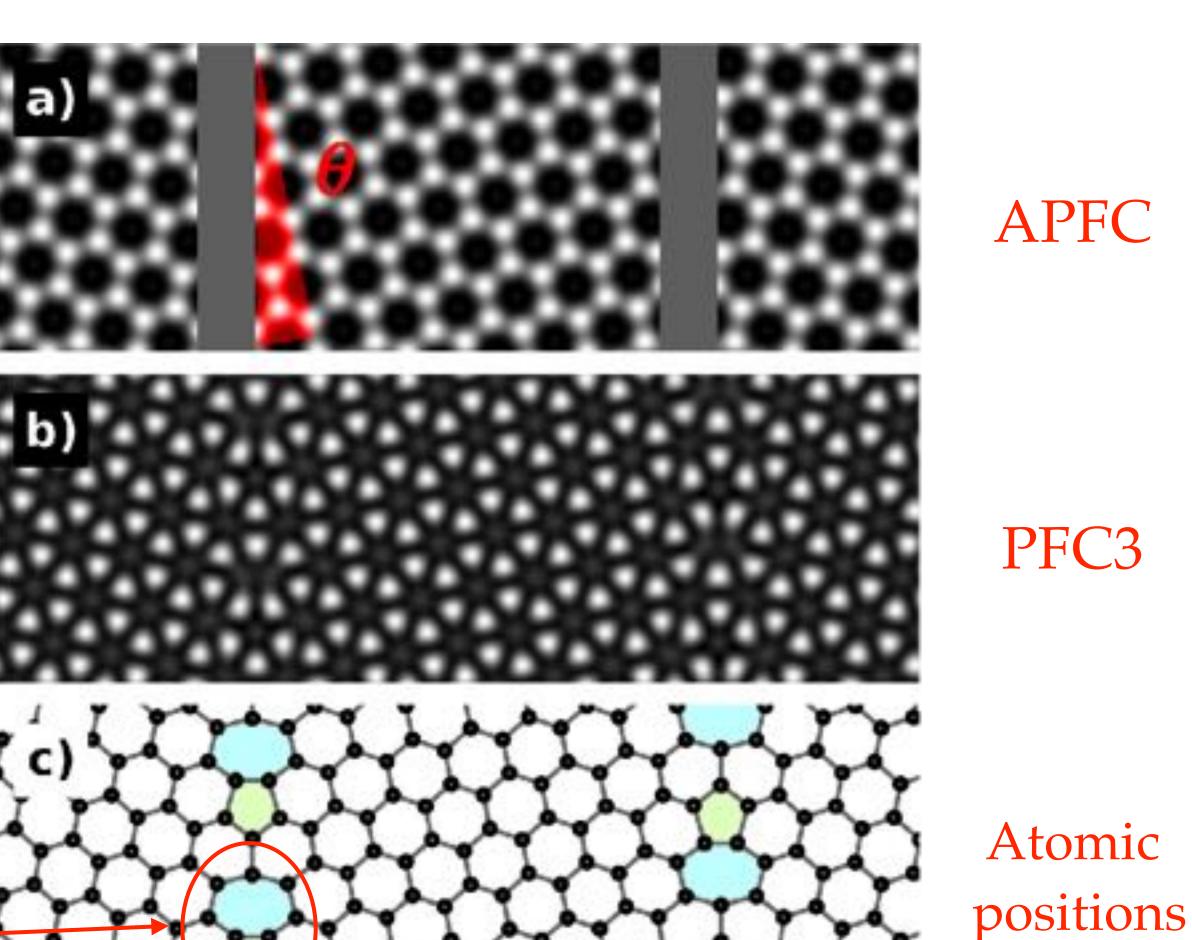




Step 1: Creating PFC Grain Boundaries

X

- * Bicrystalline layout (with two GBs) and periodic boundaries
- Finite size effects eliminated (system sizes > 10 nm)
- * Atomic positions used as input for QM-DFT and MD relaxation



4 nm

5|7

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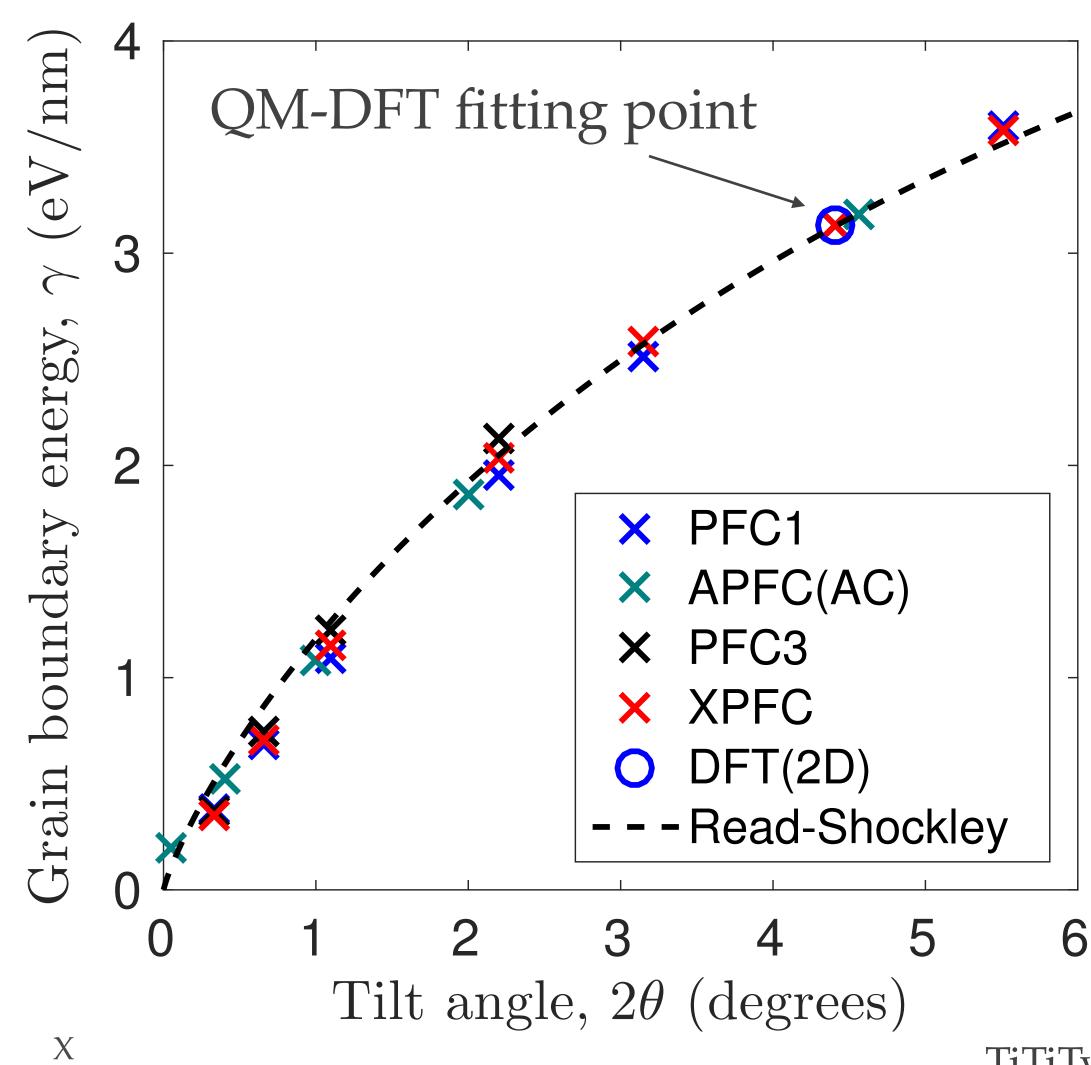




Step 2: Fitting to QM-DFT

- * Small-angle GB limit $2\theta \approx 4.6^{\circ}$ used to set energy scales in PFC models
- * Read-Shockley equation in small GB angle limit:

$$\gamma = rac{bY_{2D}}{8\pi} \theta \left(rac{3}{2} - \ln\left(2\pi\theta\right)
ight)$$





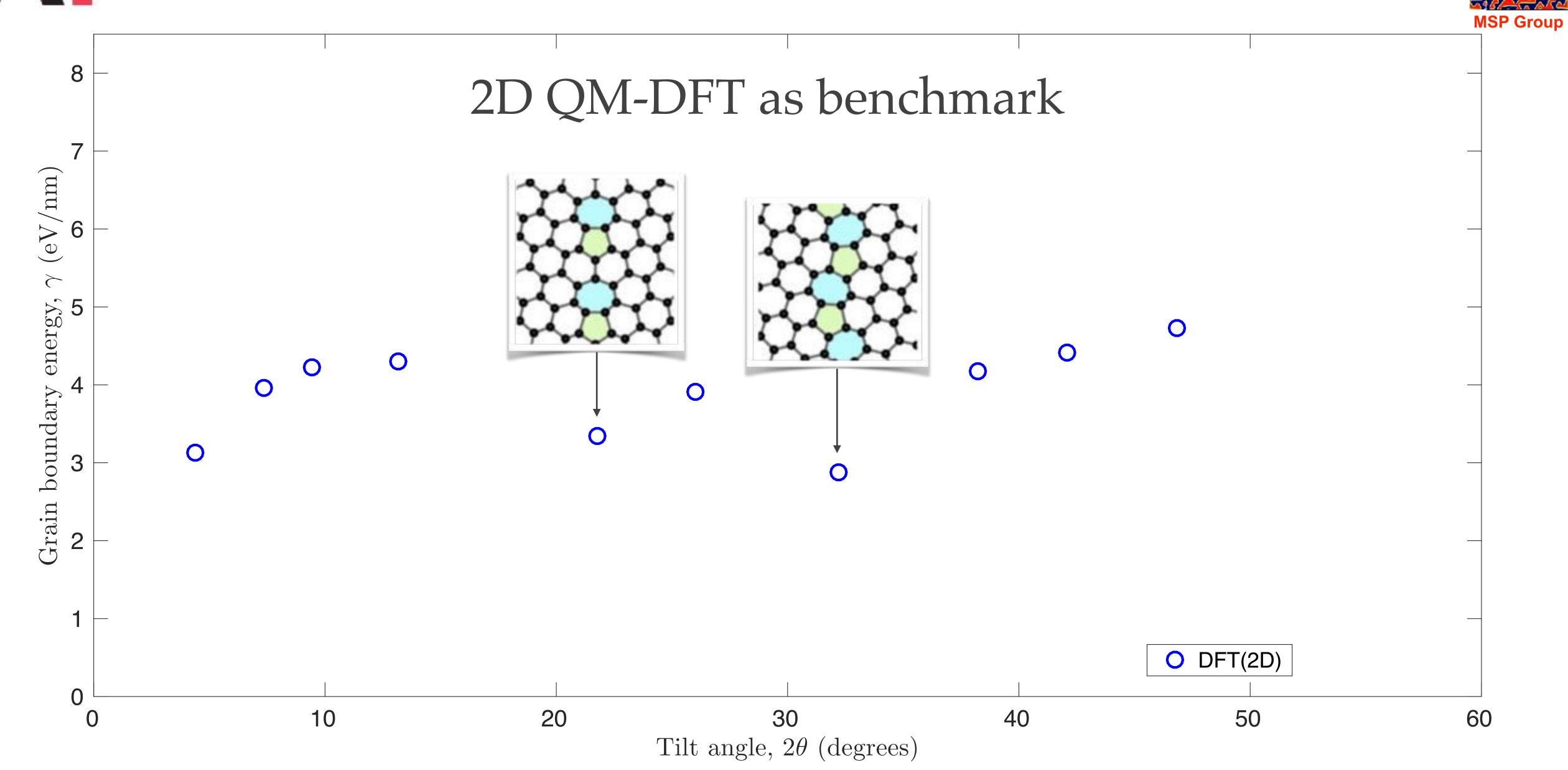


Step 3: Grain Boundary Energies



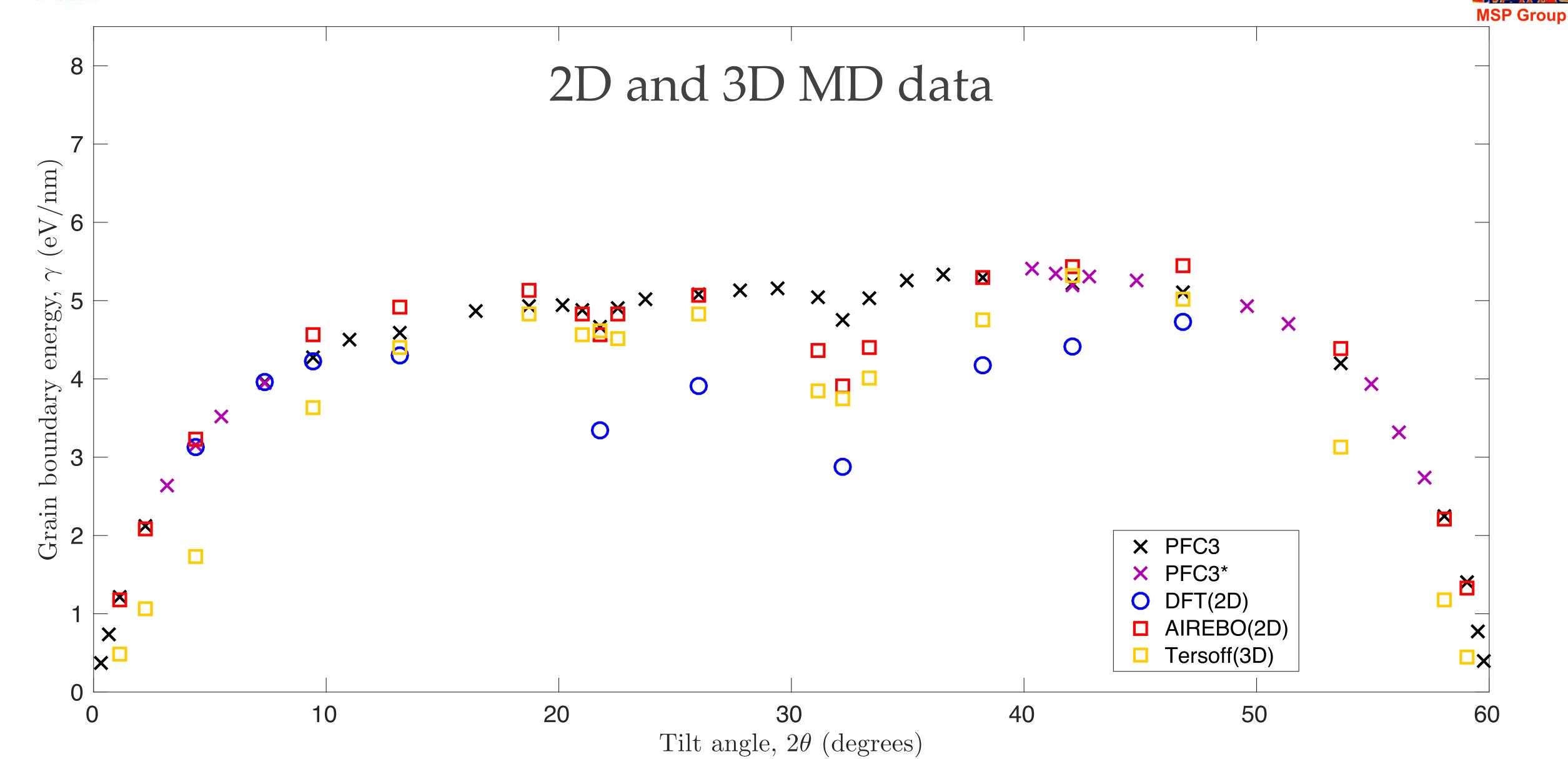


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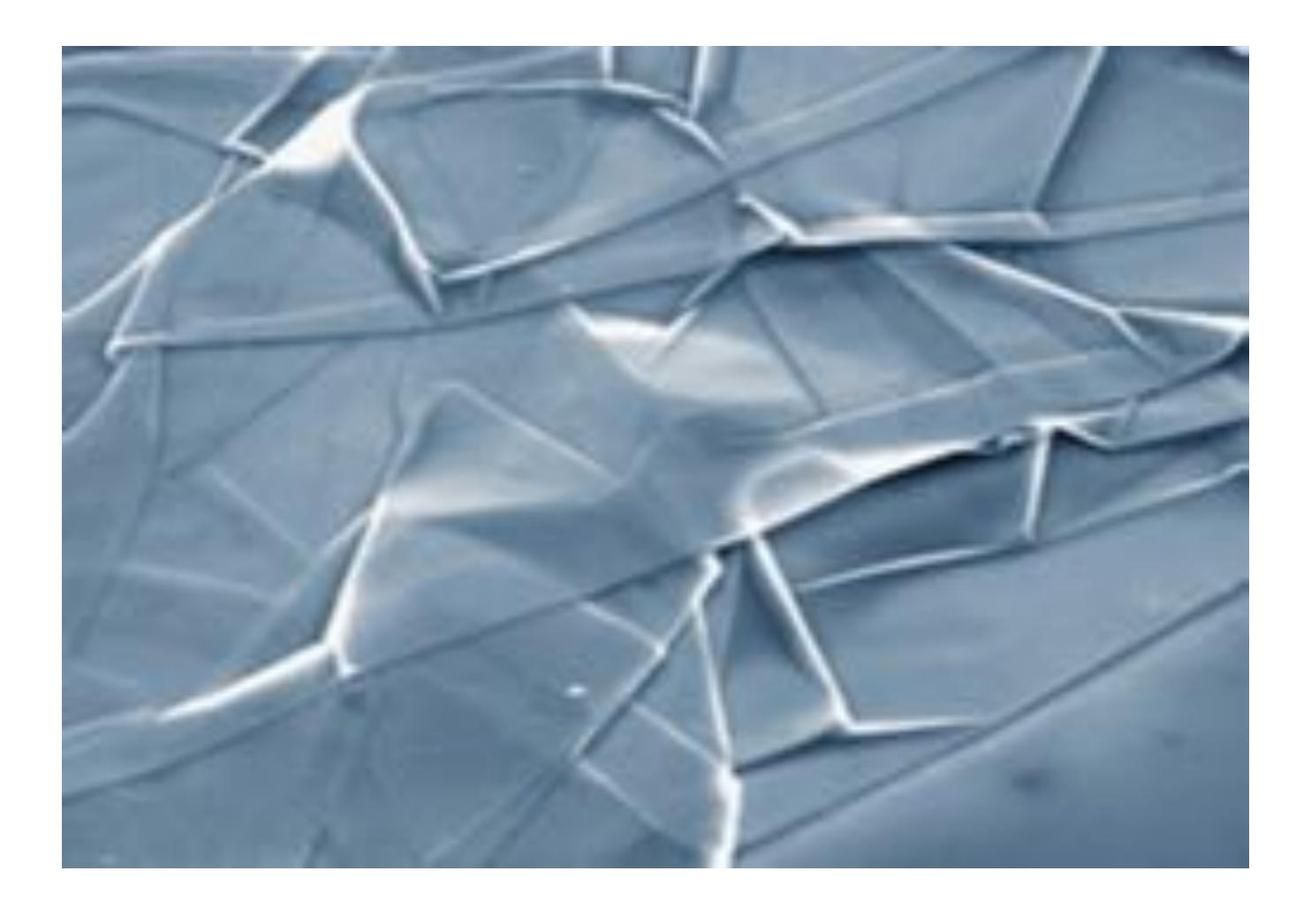


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Large Multigrain Flakes

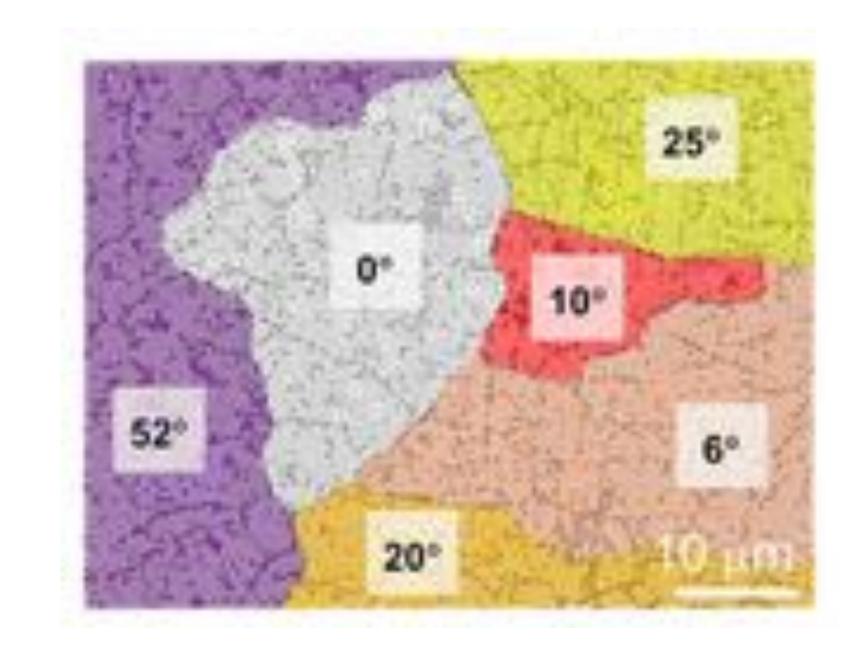






Large Multigrain Flakes

- * Large, relaxed multigrain samples can be used to study e.g. heat and electrical conduction
- * Strategy: Generate large samples with PFC1 and relax with MD (optimized Tersoff potential used here)



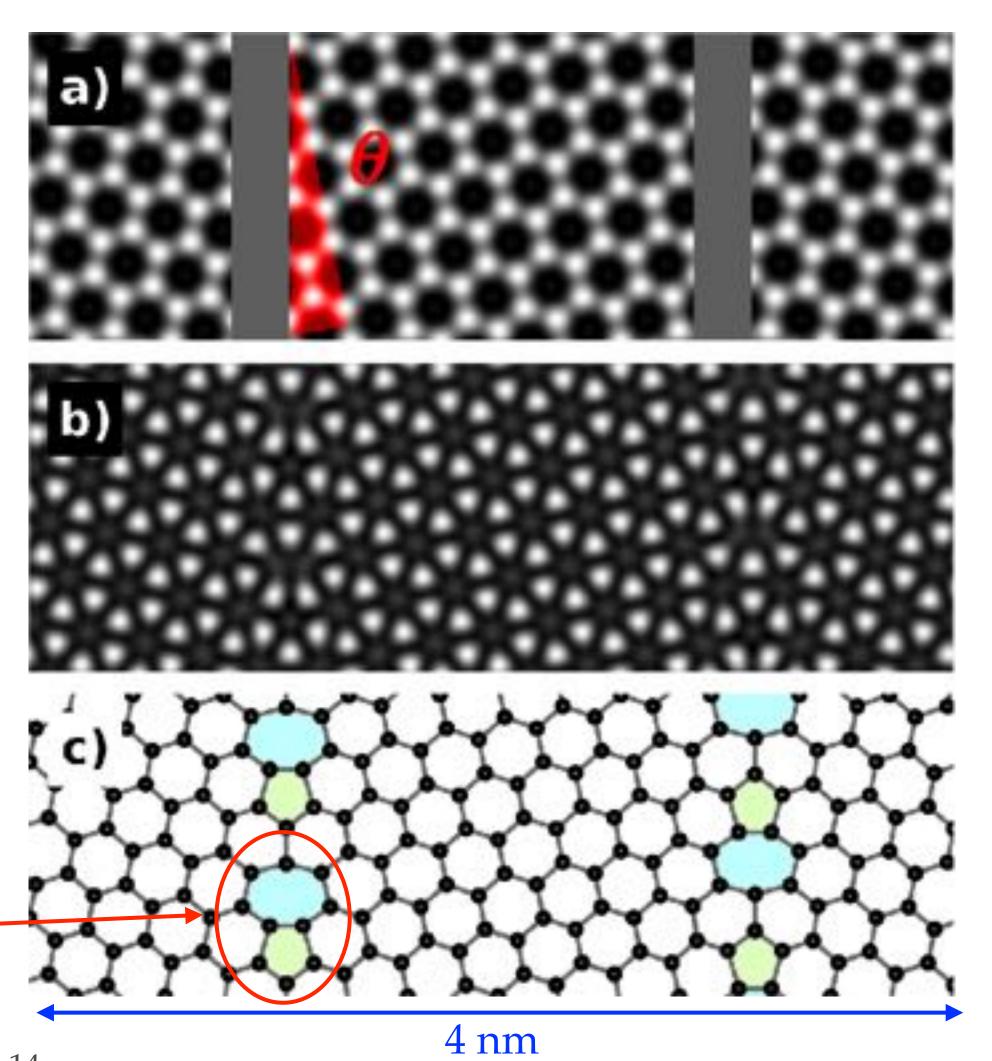
[H. Ago et al., ACS Nano 10, 3233 (2016)]





Graphene Grain Boundaries

* Bicrystalline samples are generated by having two grain boundaries with periodic boundary conditions

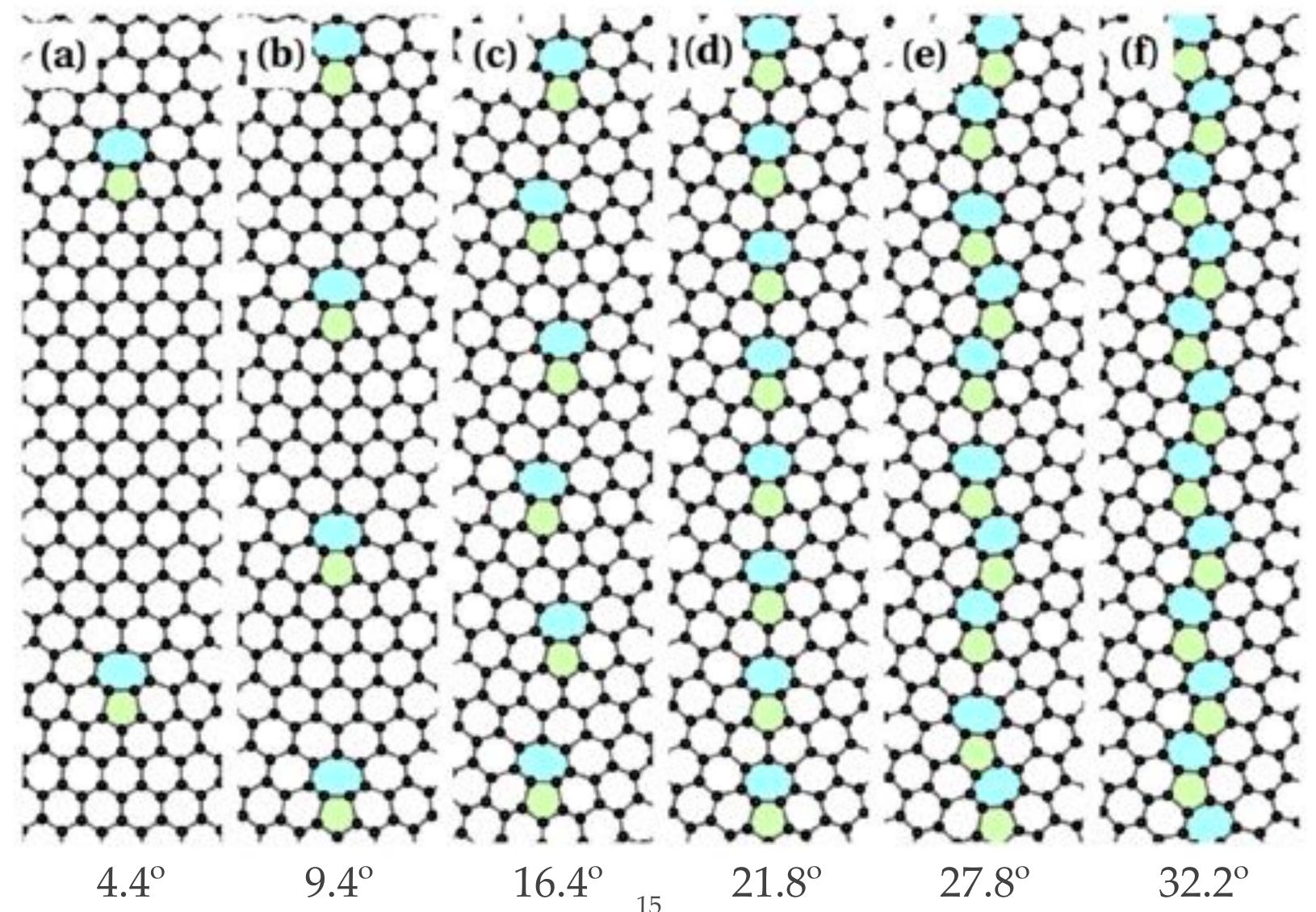


Atomic positions





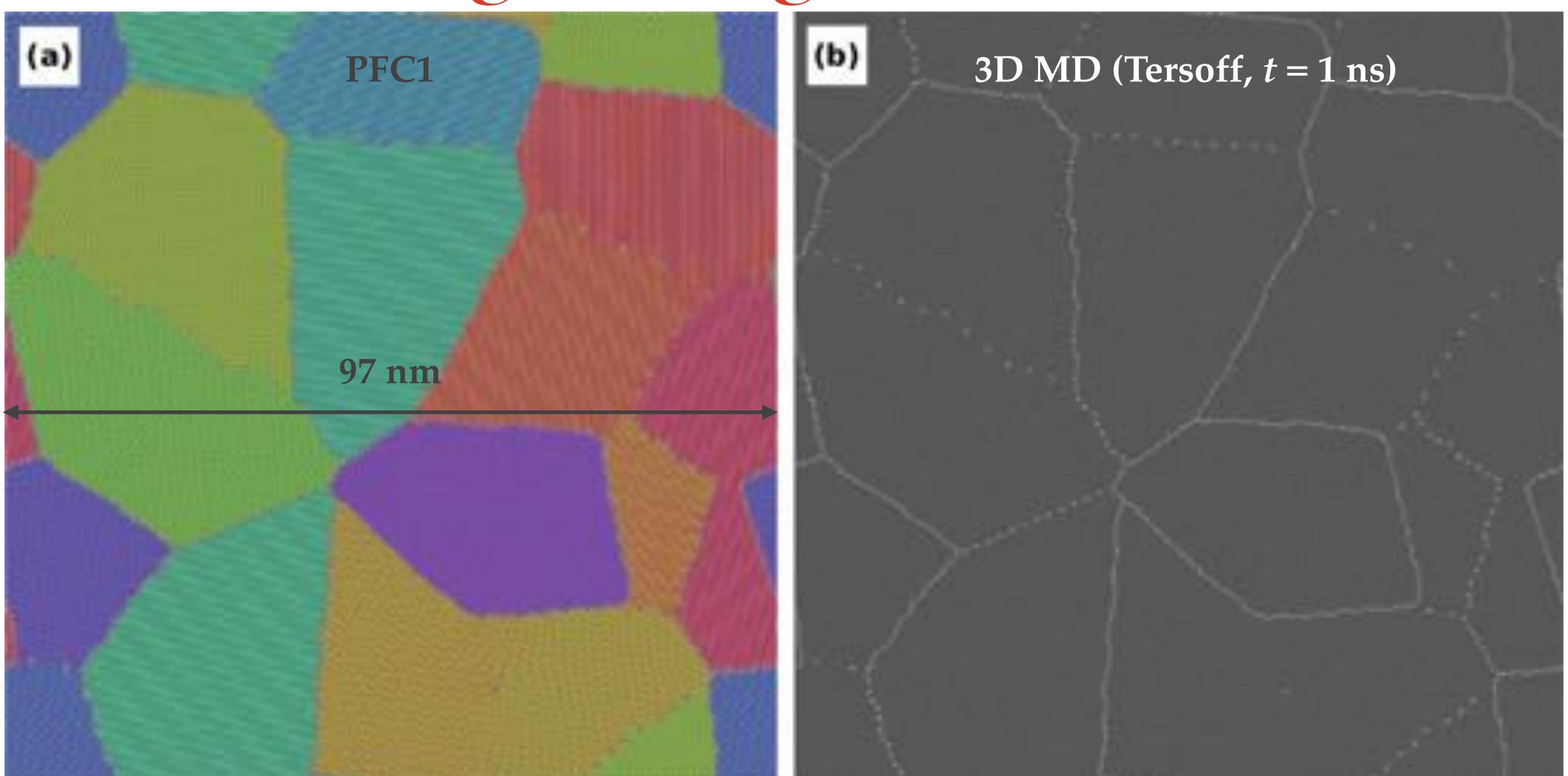
Graphene Grain Boundaries







Large Multigrain Flakes







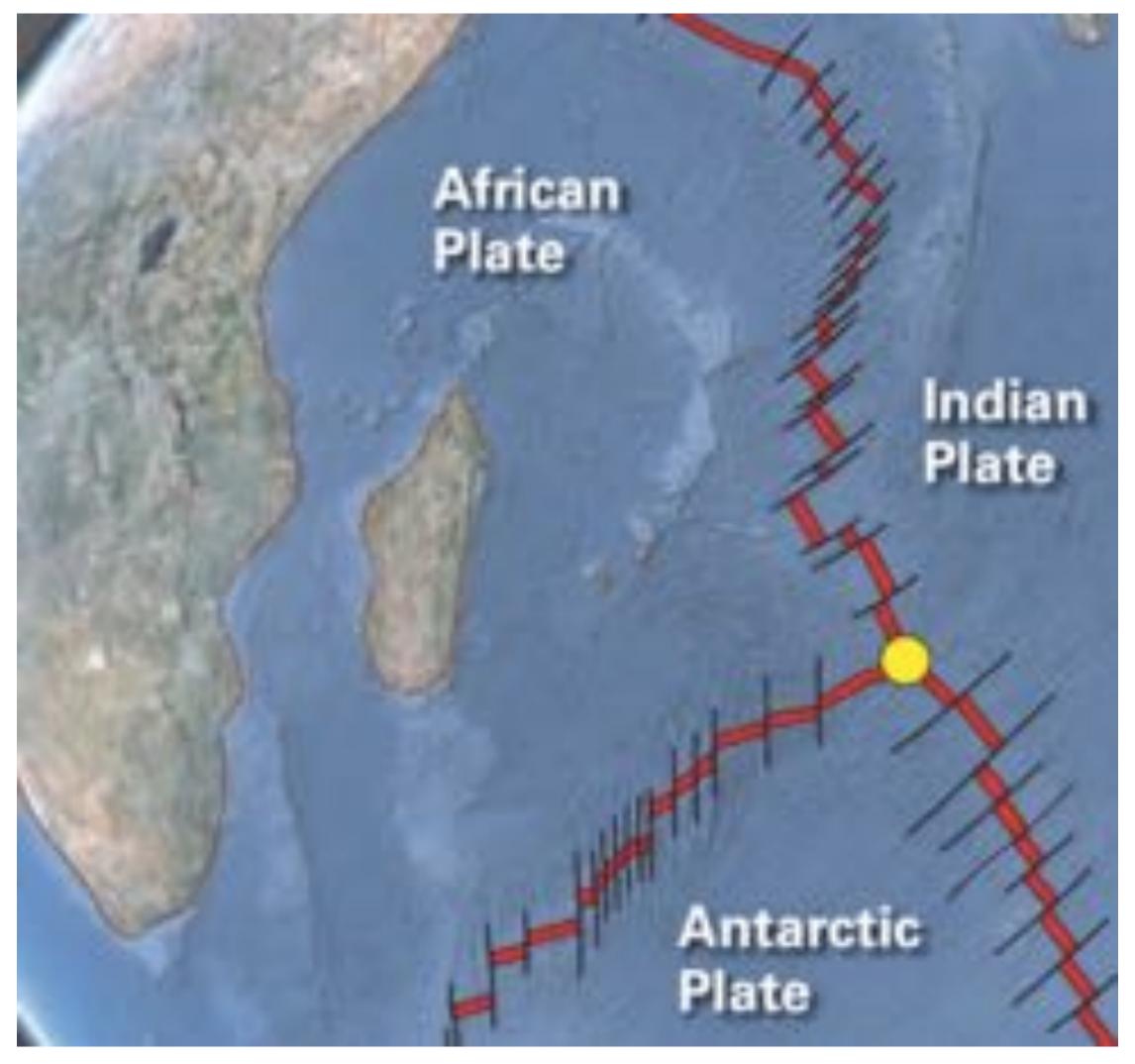
Large Multigrain Flakes





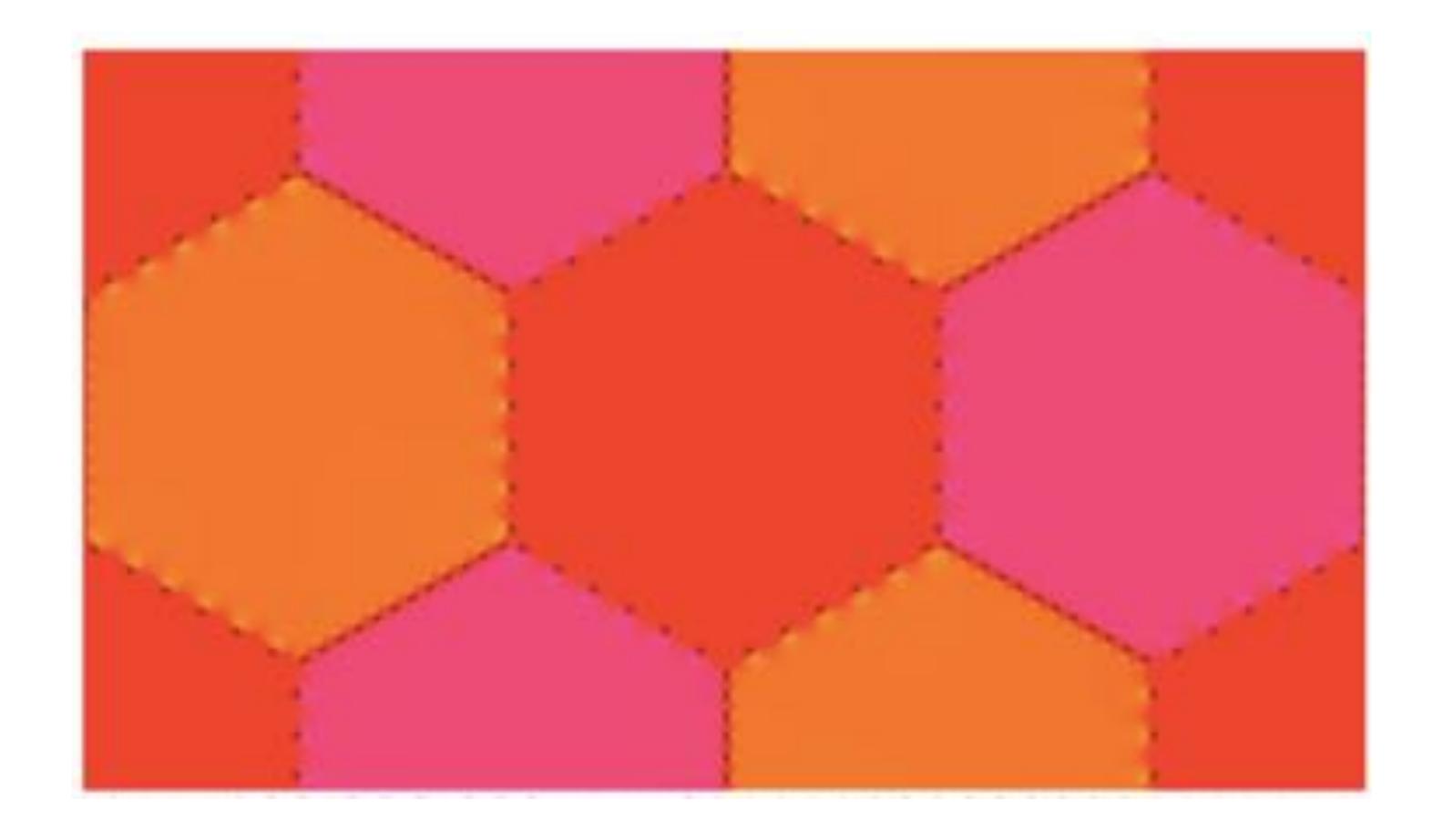


Triple Junctions



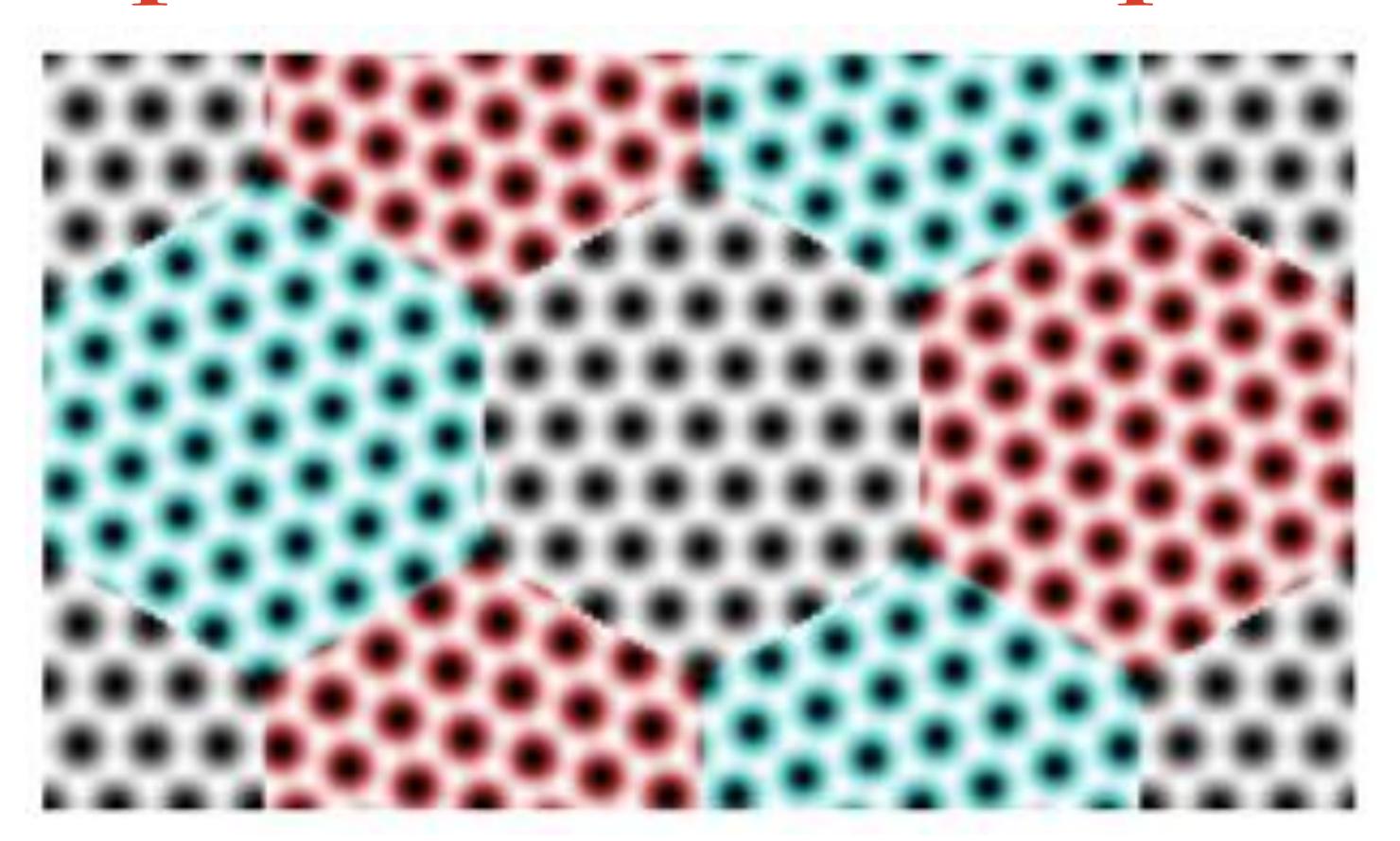








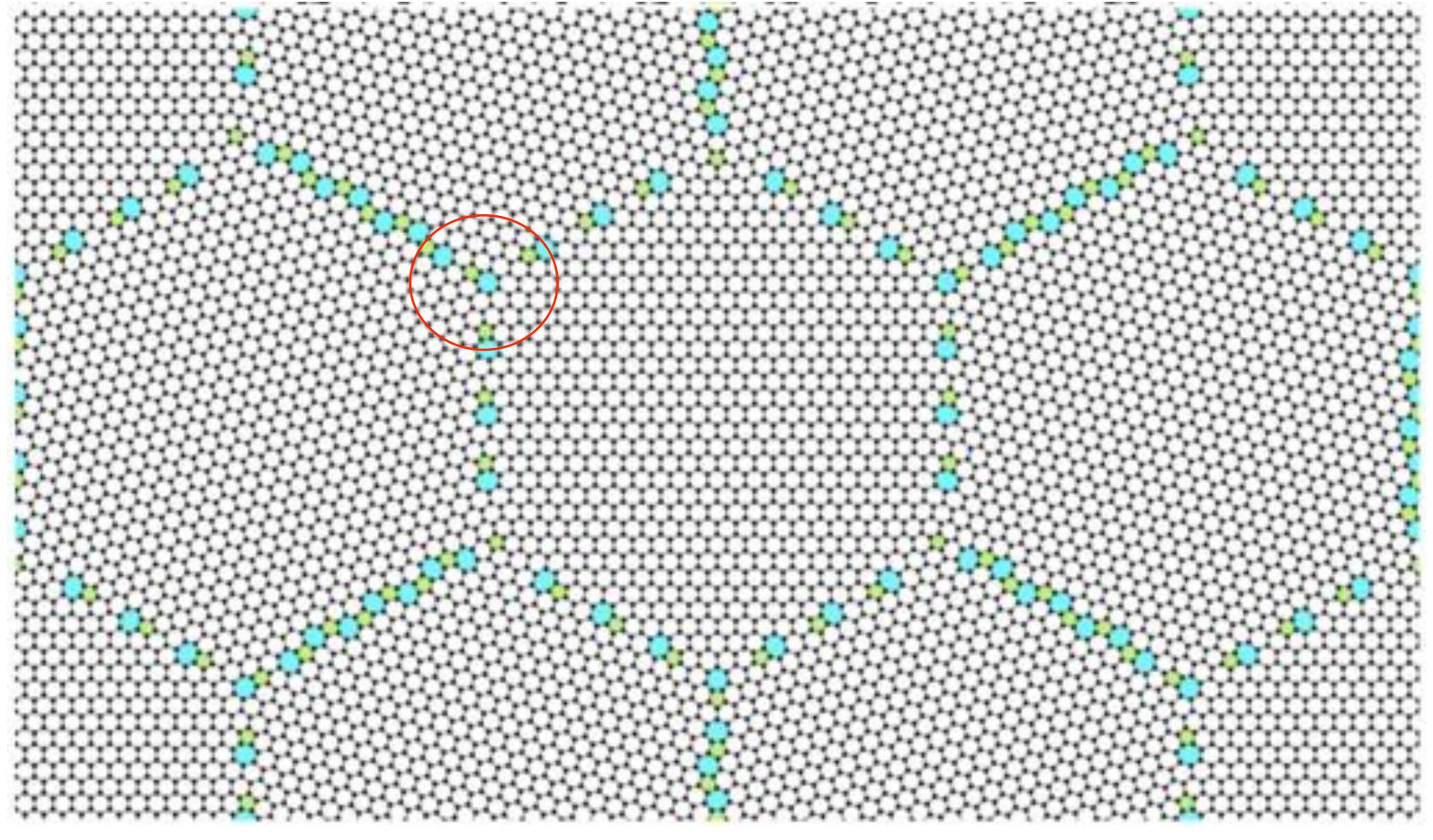




* Total energy of the system $\,F = f_{
m s} A + f_{
m L} L + f_{
m p} N$



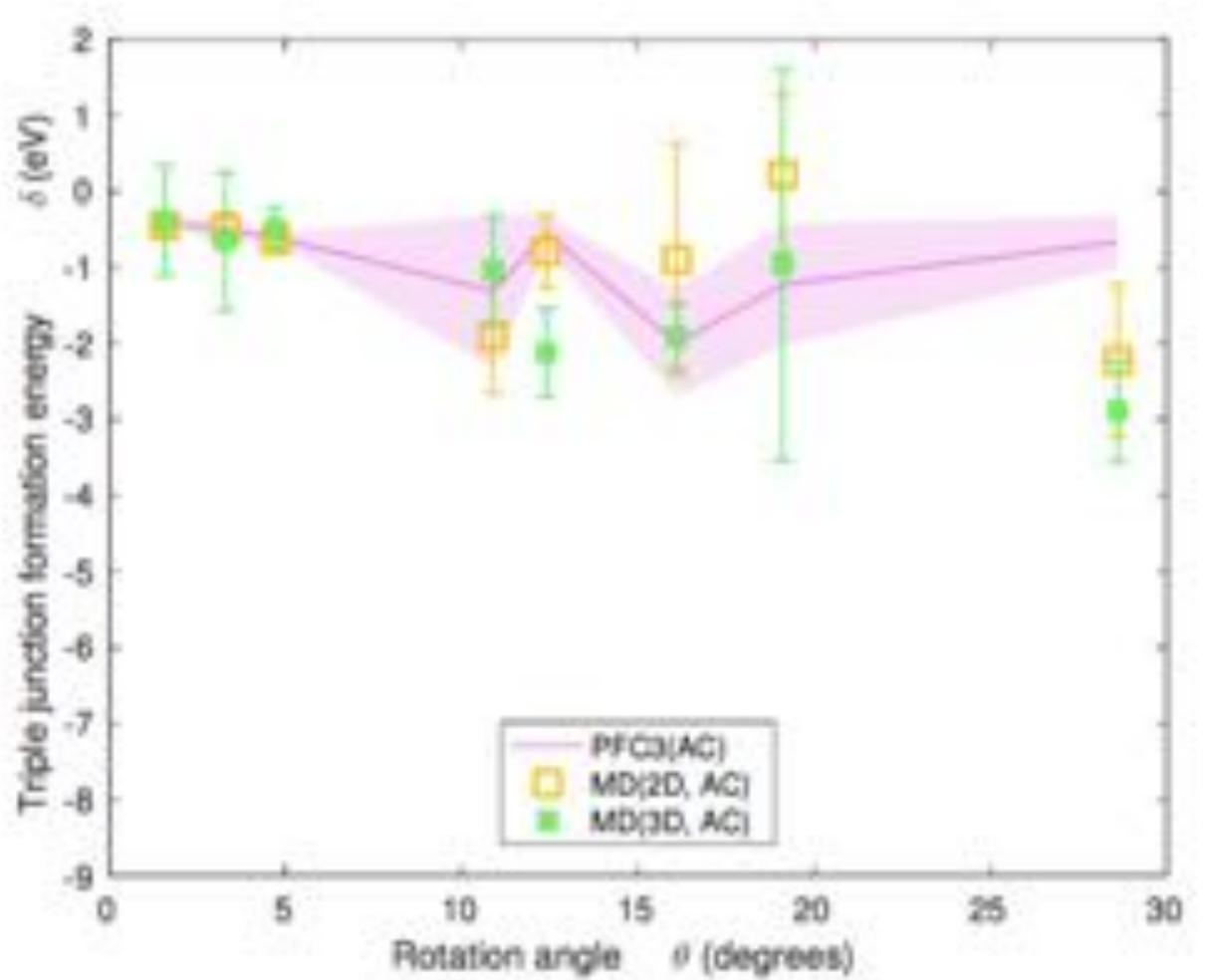




Total energy of the system $F = f_{\rm s}A + f_{\rm L}L + f_{\rm p}N$



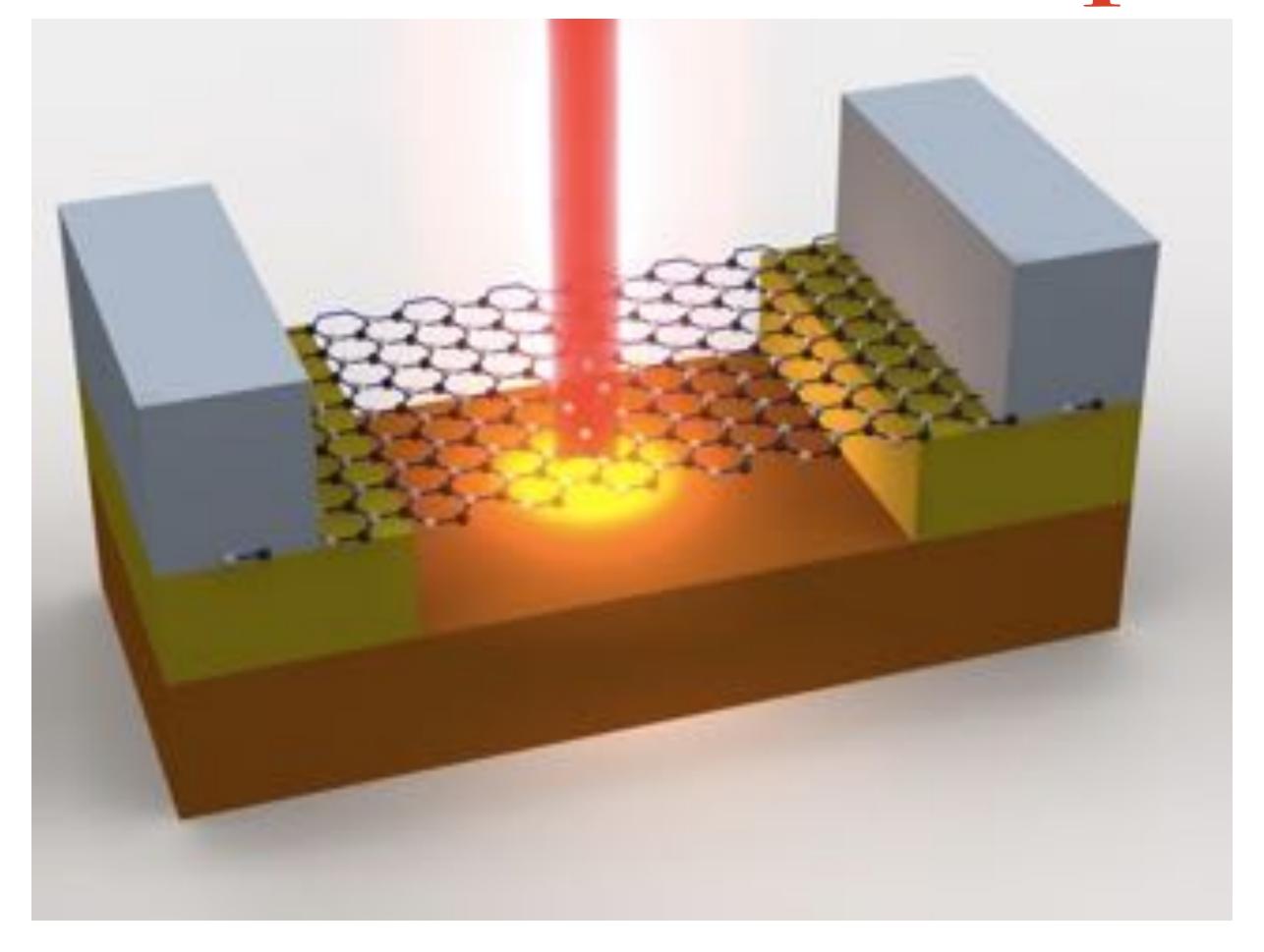




- * For most orientations, the triple junction formation energies are negative
- * Grain boundary energies are 1-5 eV/ nm i.e. (at least two) orders of magnitude larger and dominate the total energy







[Z. Fan, L.F.C. Pereira, P. Hirvonen, M.M. Ervasti, K.R. Elder, D. Donadio, T. Ala-Nissila, and A. Harju, Phys. Rev. B **95**, 144309 (2017); Nano Letters 7b1072 (2017); K. Azizi, P. Hirvonen, Z. Fan, A. Harju, K.R. Elder, T. Ala-Nissila, and S. M. Vaez-Allaei, Carbon **125**, 384 (2017)]





* Heat conductivity in and out-of plane can be calculated from the heat (energy) flow autocorrelation function

$$\kappa_{\mu\nu}(t) = rac{1}{k_B T^2 V} \int_0^t dt' C_{\mu\nu}(t')$$

where

$$C_{\mu\nu}(t) = \langle J_{\mu}(0)J_{\nu}(t)\rangle$$





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where

$$C_{\mu
u}^{
m in/out}$$
 in/out in/out $C_{\mu
u}^{
m in/out}$ in/out $C_{\mu
u}$

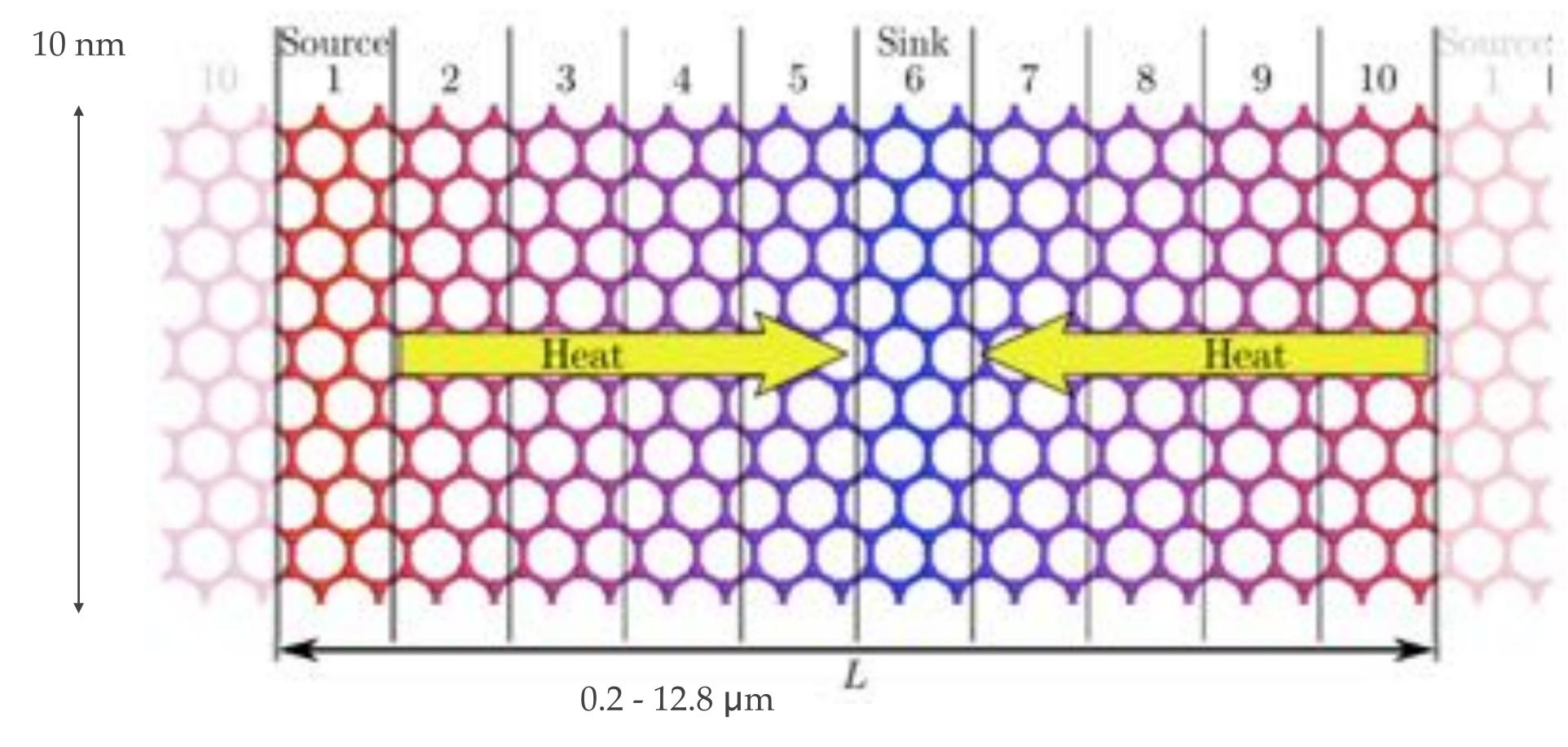
and

$$m{J}^{ ext{in}} = \sum_{i} \sum_{j \neq i} m{r}_{ij} \left(rac{\partial U_j}{\partial x_{ji}} v_{xi} + rac{\partial U_j}{\partial y_{ji}} v_{yi}
ight), \, m{J}^{ ext{out}} = \sum_{i} \sum_{j \neq i} m{r}_{ij} \left(rac{\partial U_j}{\partial z_{ji}} v_{zi}
ight)$$





* ... or from Fourier's law by setting an external thermal gradient

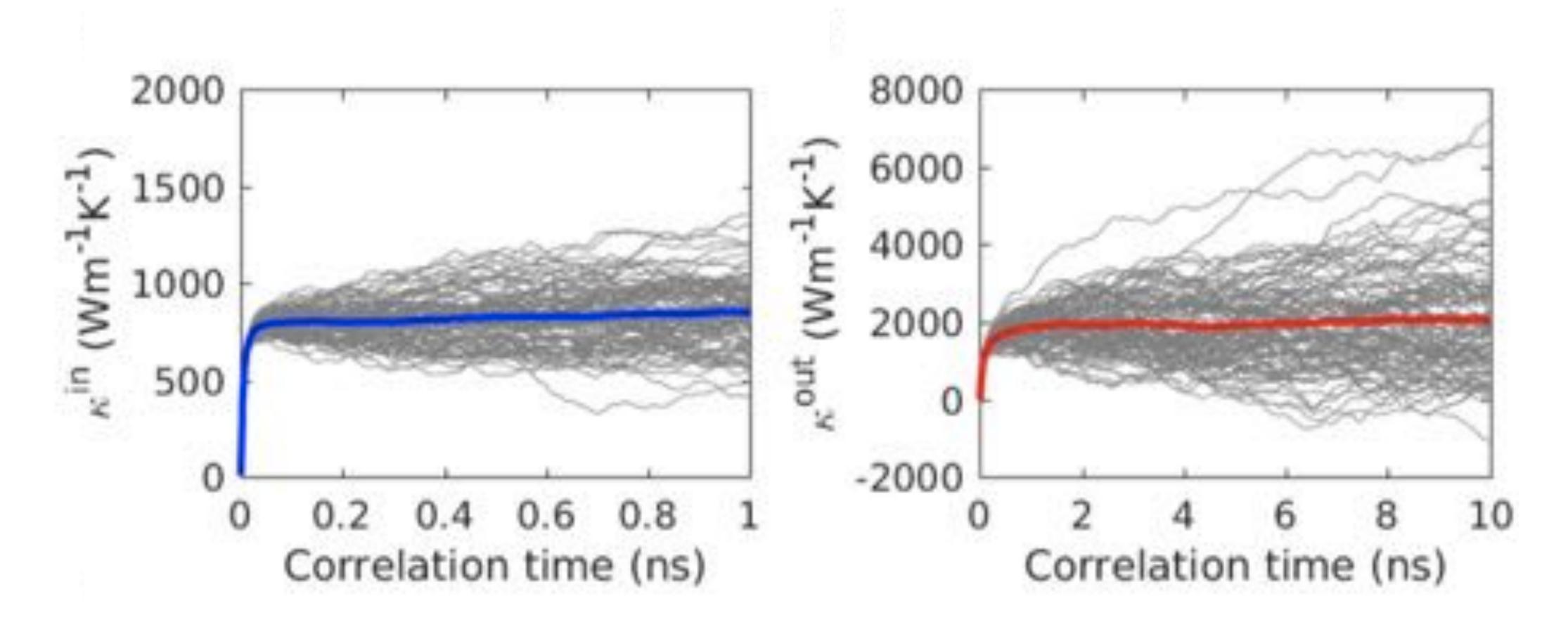


$$\kappa(L) = rac{Q^{
m ext}/2}{S|\nabla T|}$$





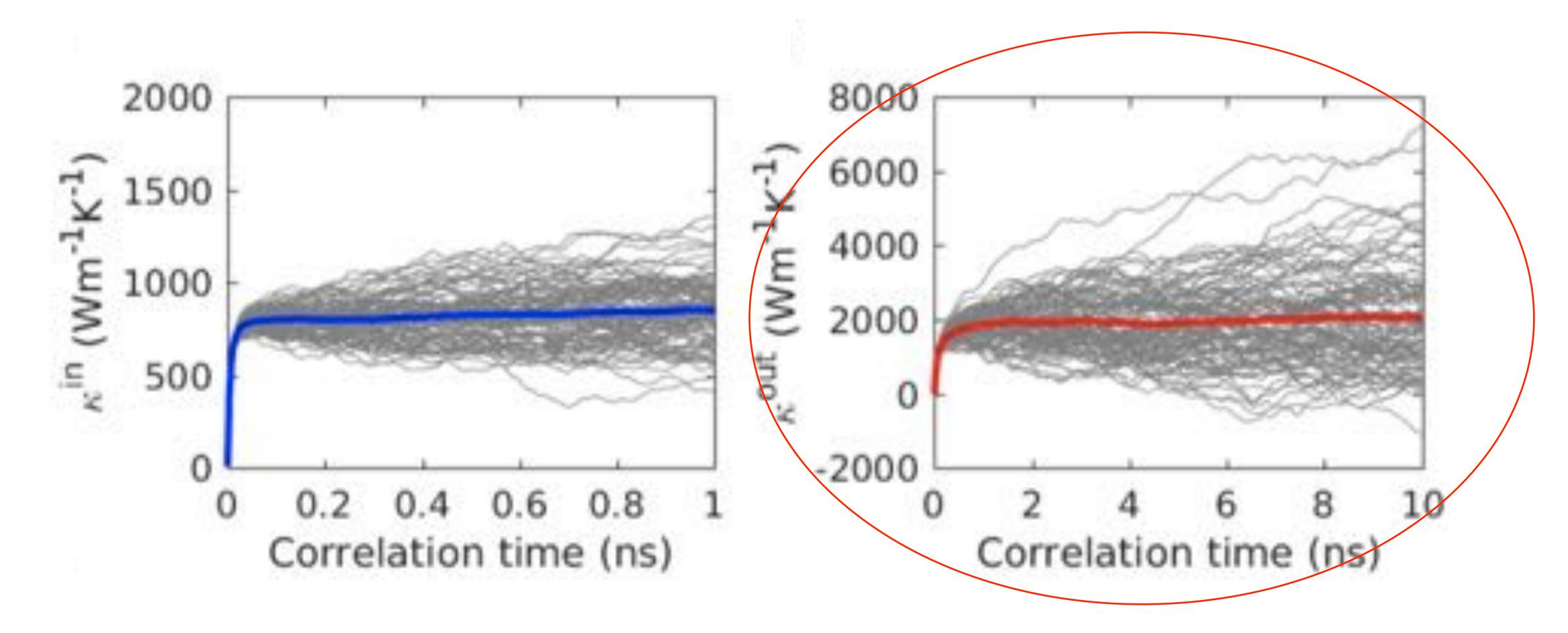
* The out-of-plane component converges much slower than the in-plane one







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* For pristine graphene we find at T = 300 K (MD with opt. Tersoff)

$$\kappa_0^{\rm in} \approx 800 \ {\rm Wm^{-1} K^{-1}} \ {\rm and} \ \kappa_0^{\rm out} \approx 2 \ 100 \ {\rm Wm^{-1} K^{-1}}$$

$$\kappa_0 = \kappa_0^{\text{in}} + \kappa_0^{\text{out}} = 2 \ 900 \ \text{Wm}^{-1} \text{K}^{-1}$$

Experimentally $\kappa_0 \approx 1500 - 2500 \; \mathrm{Wm}^{-1} \mathrm{K}^{-1}$





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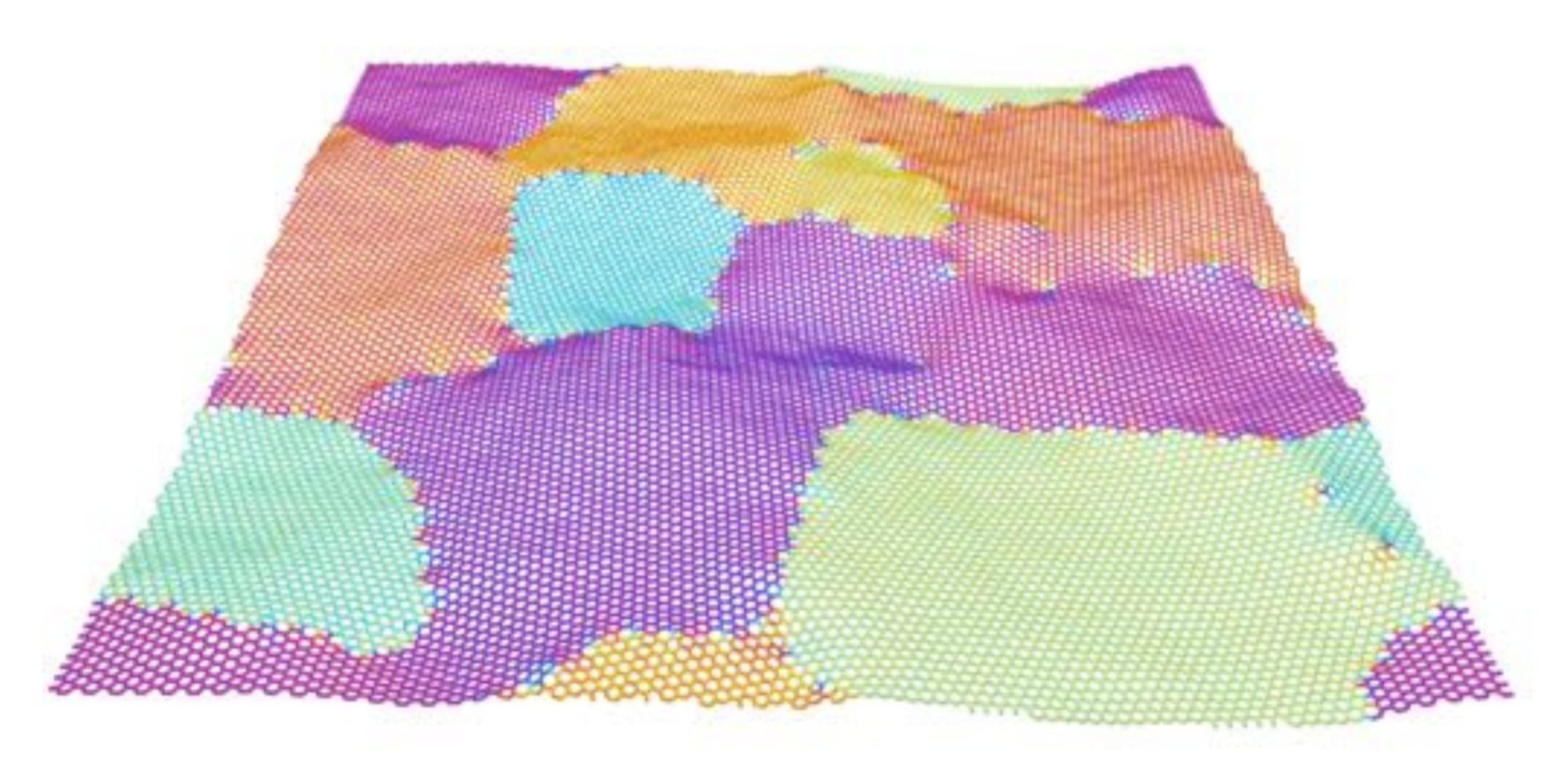
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Experimentally $\kappa_0 \approx 1500 - 2500 \; \mathrm{Wm}^{-1} \mathrm{K}^{-1}$

* For uniaxially strained graphene the out-of-plane component diverges for 2% strain





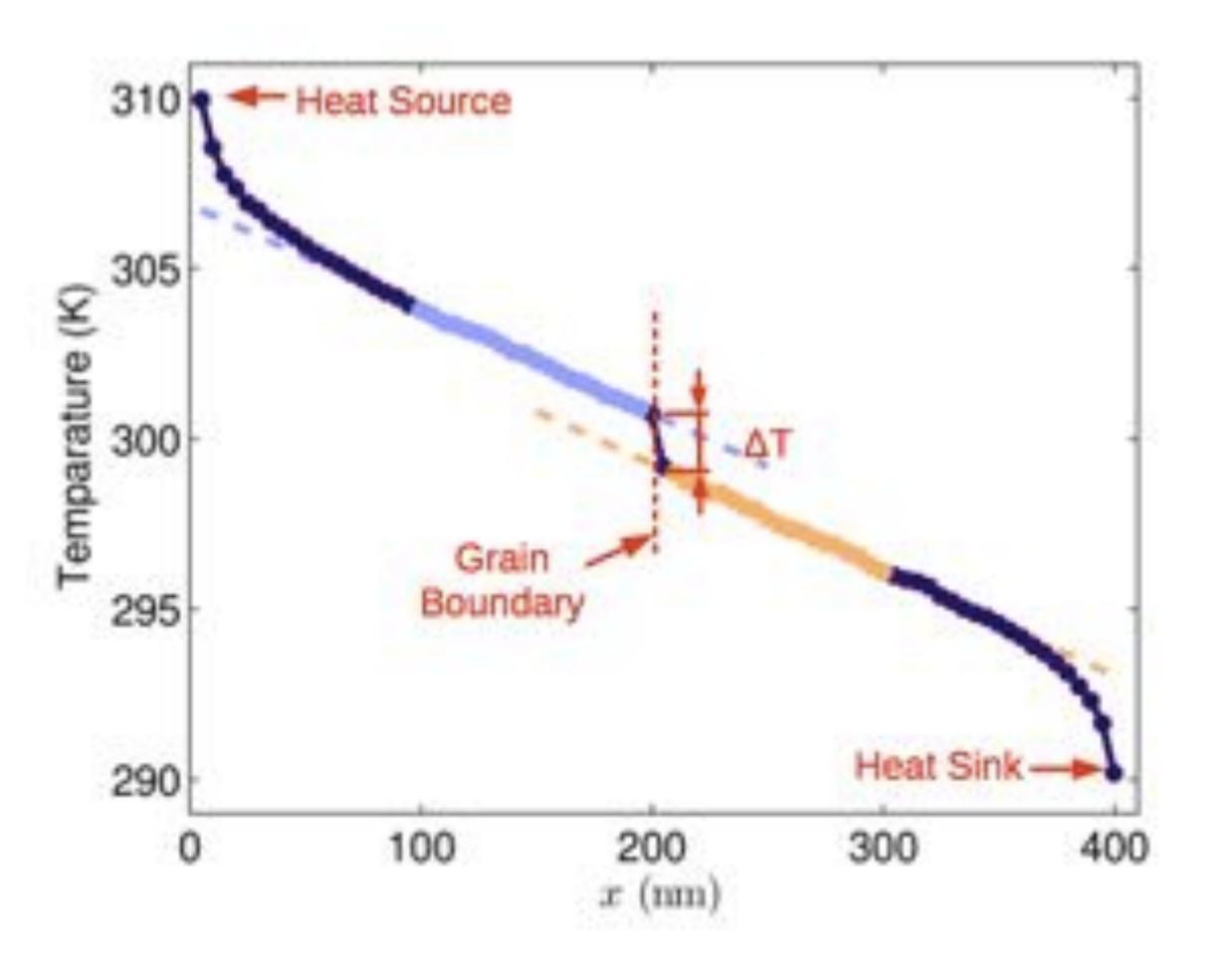






* Heat flow across an interface is characterised by *Kapitza conductance*

$$Q = G\Delta T$$





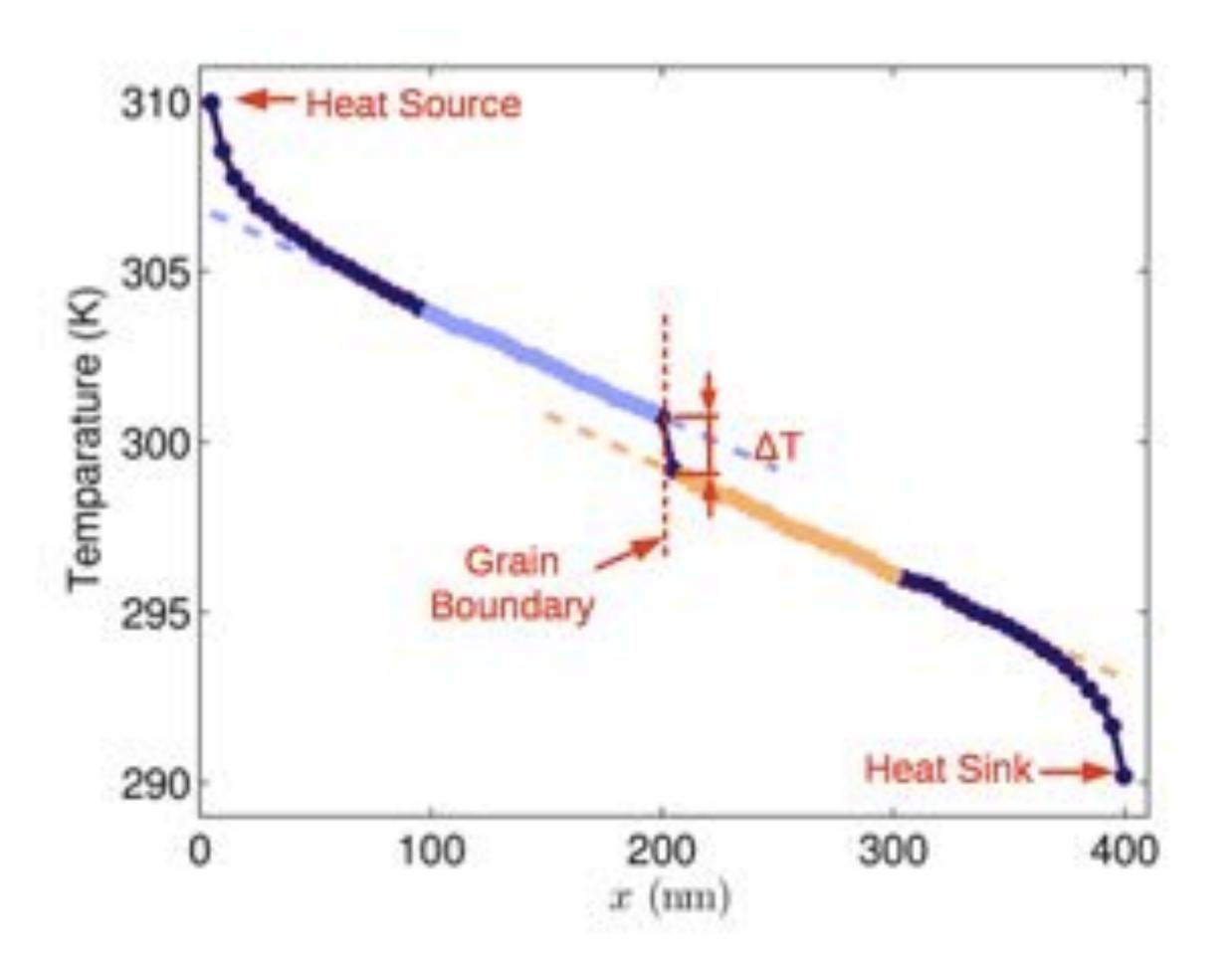


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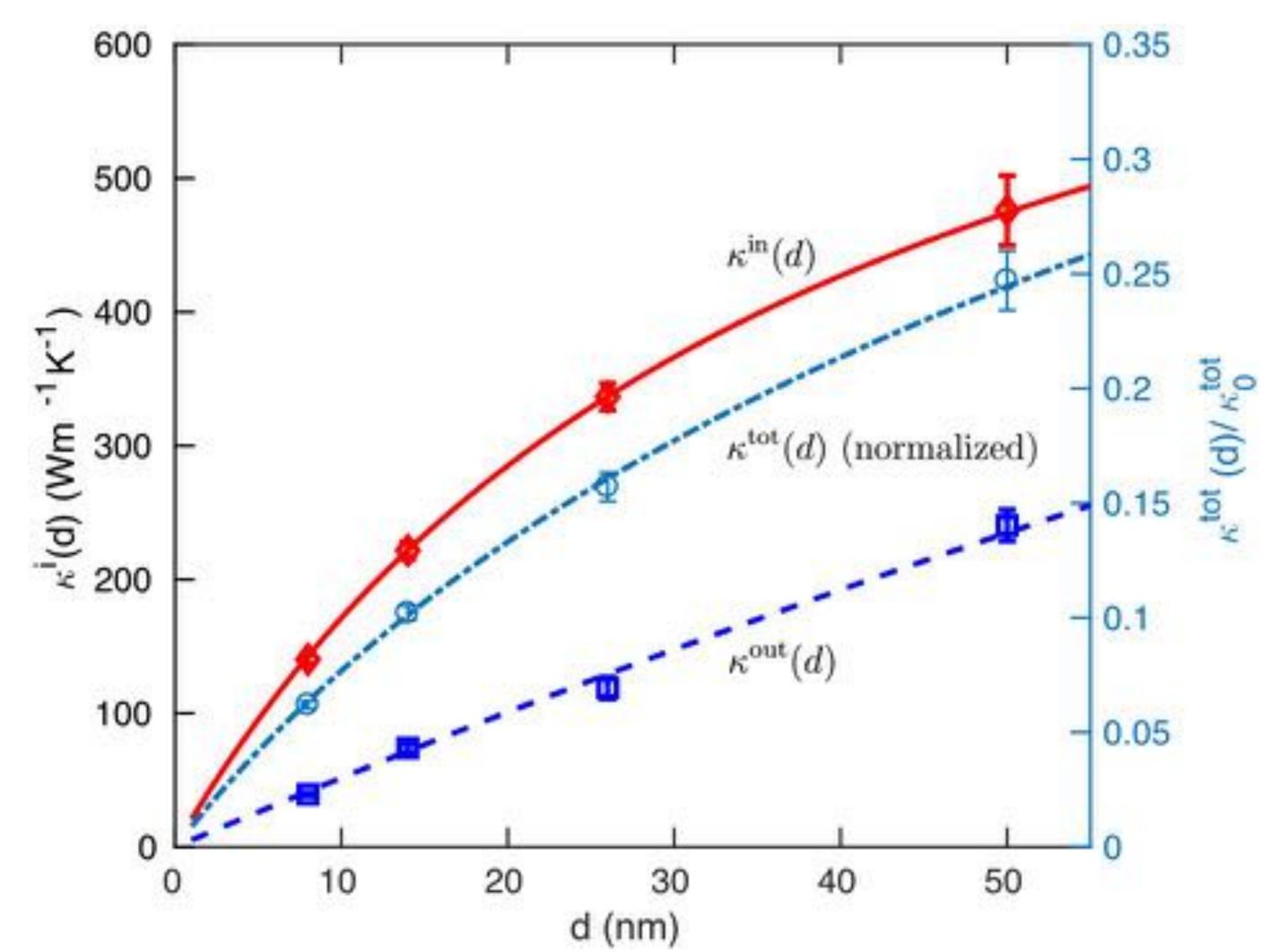
* The *Kapitza length L* equals the thickness of the material of thermal conductivity κ that provides the same change in temperature as a given interface

$$L^i = \kappa^i/G^i$$
 $(i = \text{in, out})$





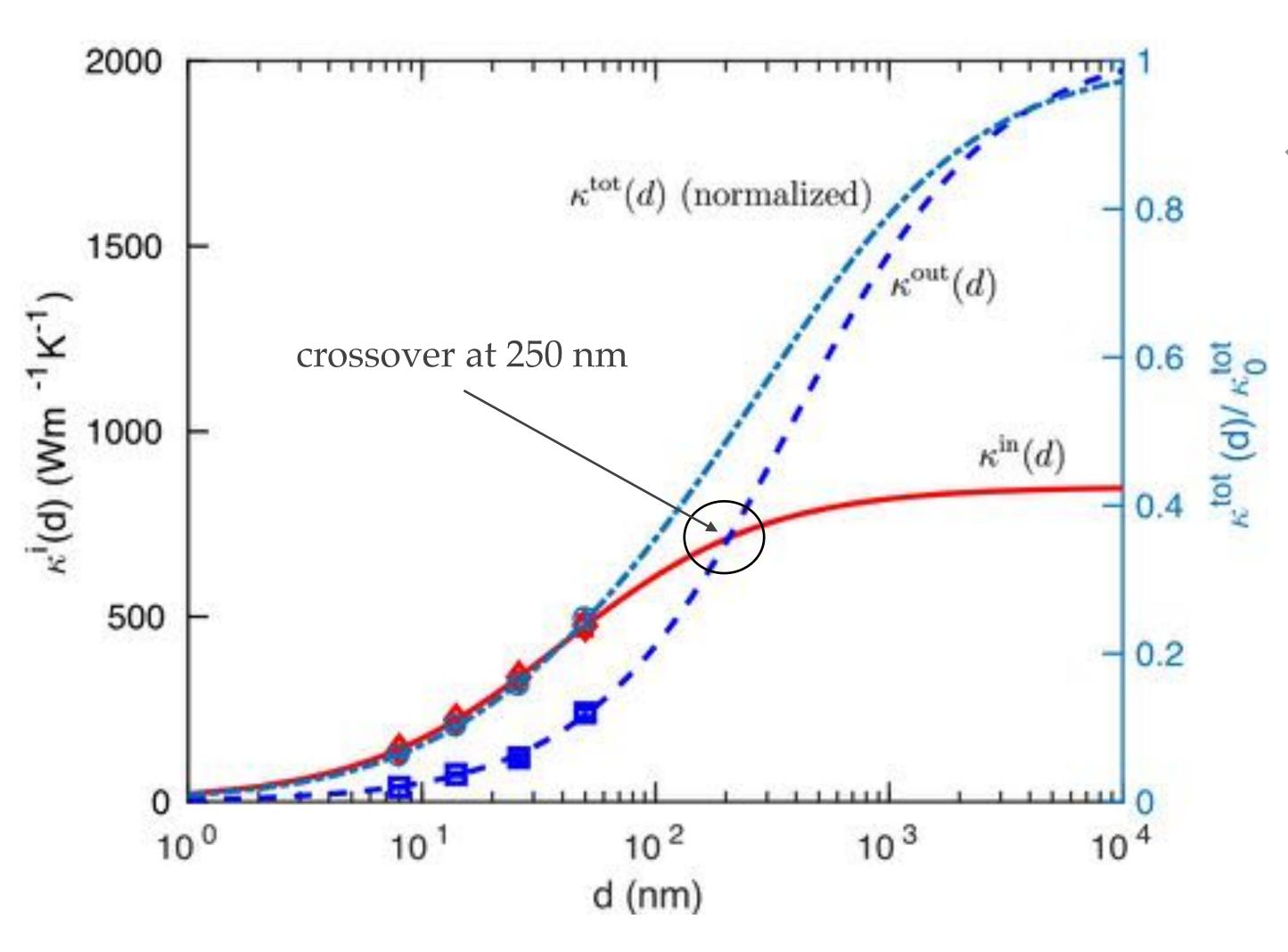




* Heat conductivity out of plane is *strongly suppressed* by grains







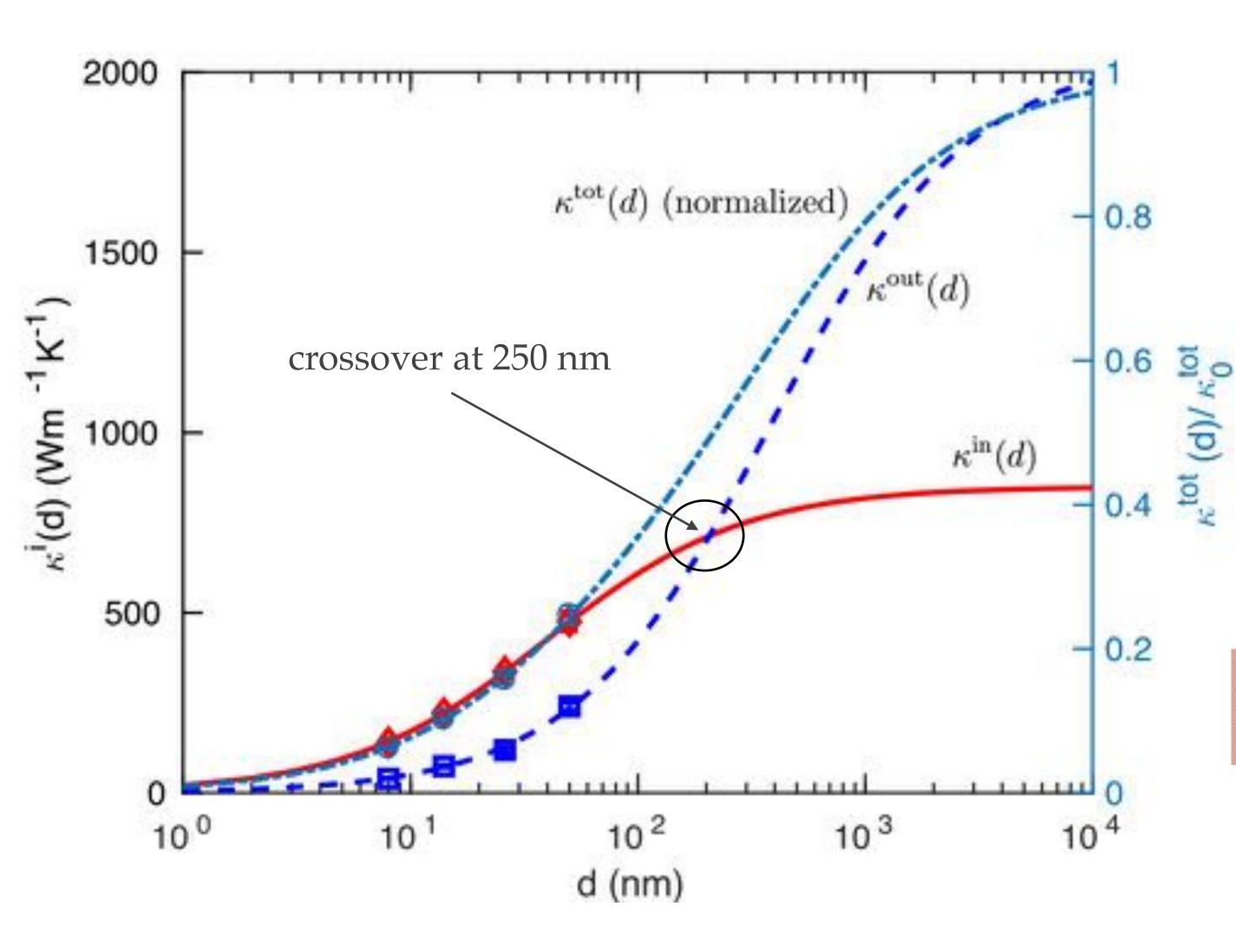
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Heat Conduction in Multigrain Flakes



- * Heat conductivity out of plane is *strongly suppressed* by grains
- * The Kapitza length *L* for out of plane is an *order of magnitude larger* than that of the in-plane

 $L^{\rm out} \approx 400 \text{ nm} \gg L^{\rm in} \approx 40 \text{ nm}$













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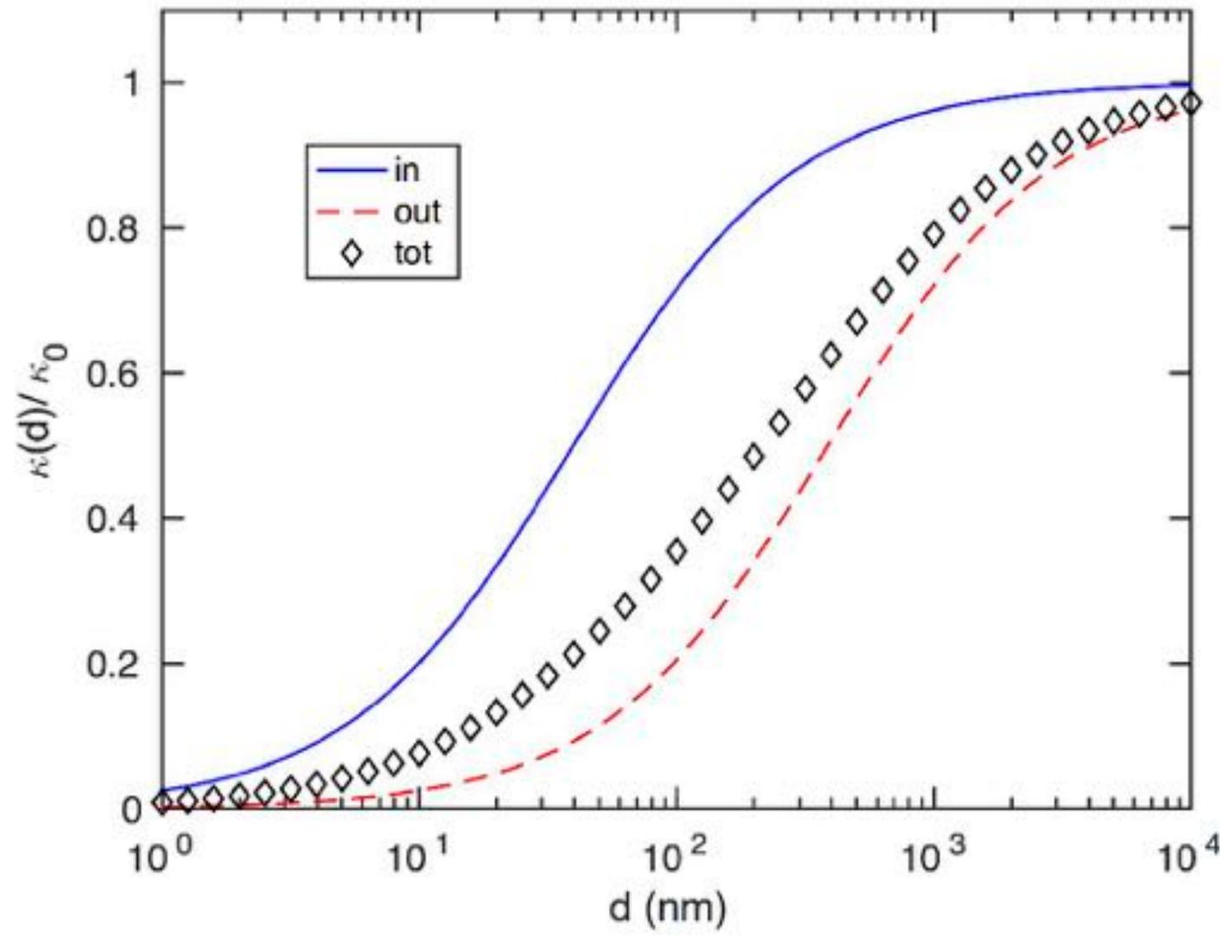
DOb 10.3038/ncomms14486 OPEN

Tailoring the thermal and electrical transport properties of graphene films by grain size engineering

Teng Ma¹, Zhibo Liu¹, Jinxiu Wen², Yang Gao¹, Xibiao Ren³, Huanjun Chen², Chuanhong Jin³, Xiu-Liang Ma¹, Ningsheng Xu², Hui-Ming Cheng¹ & Wencai Ren¹

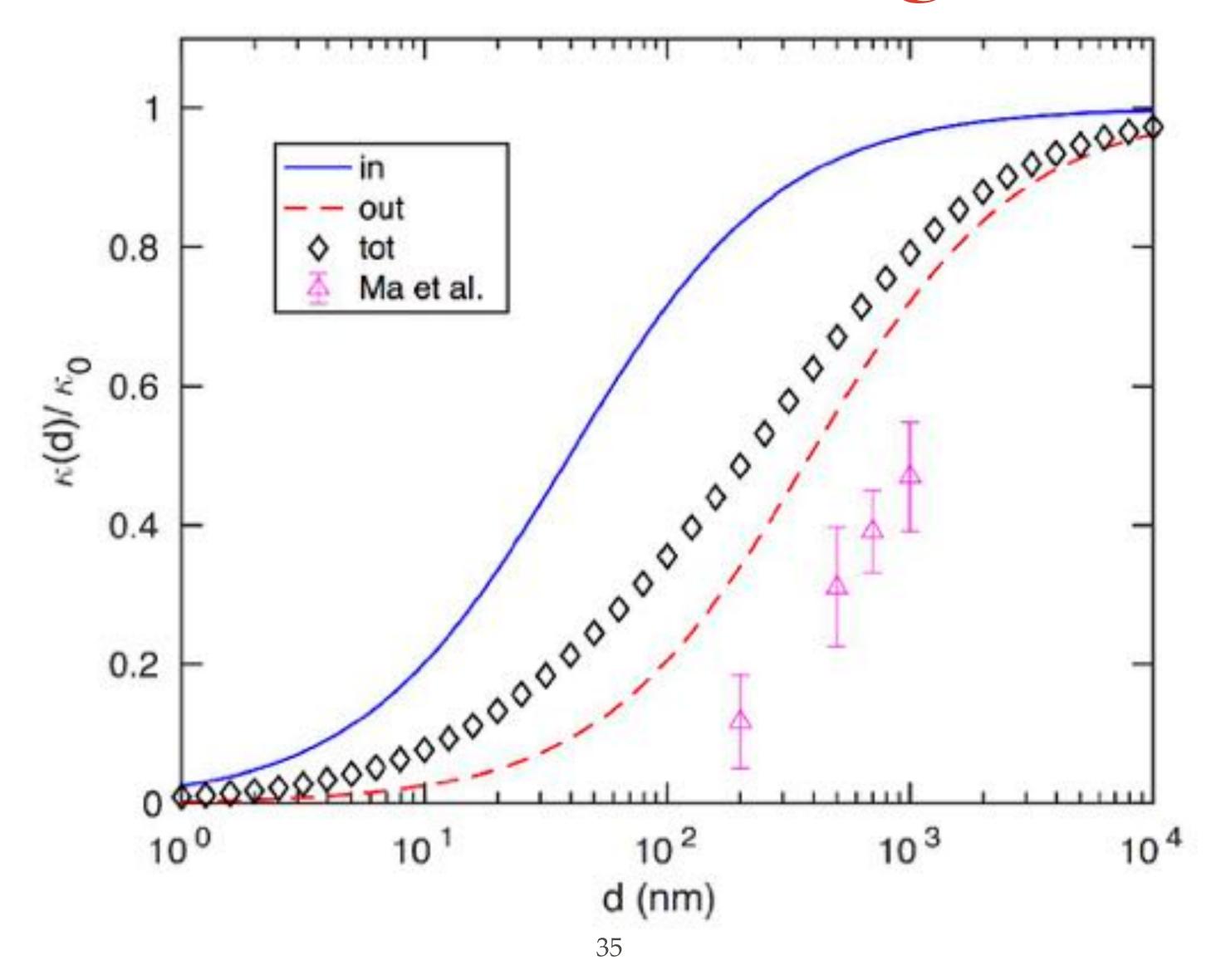










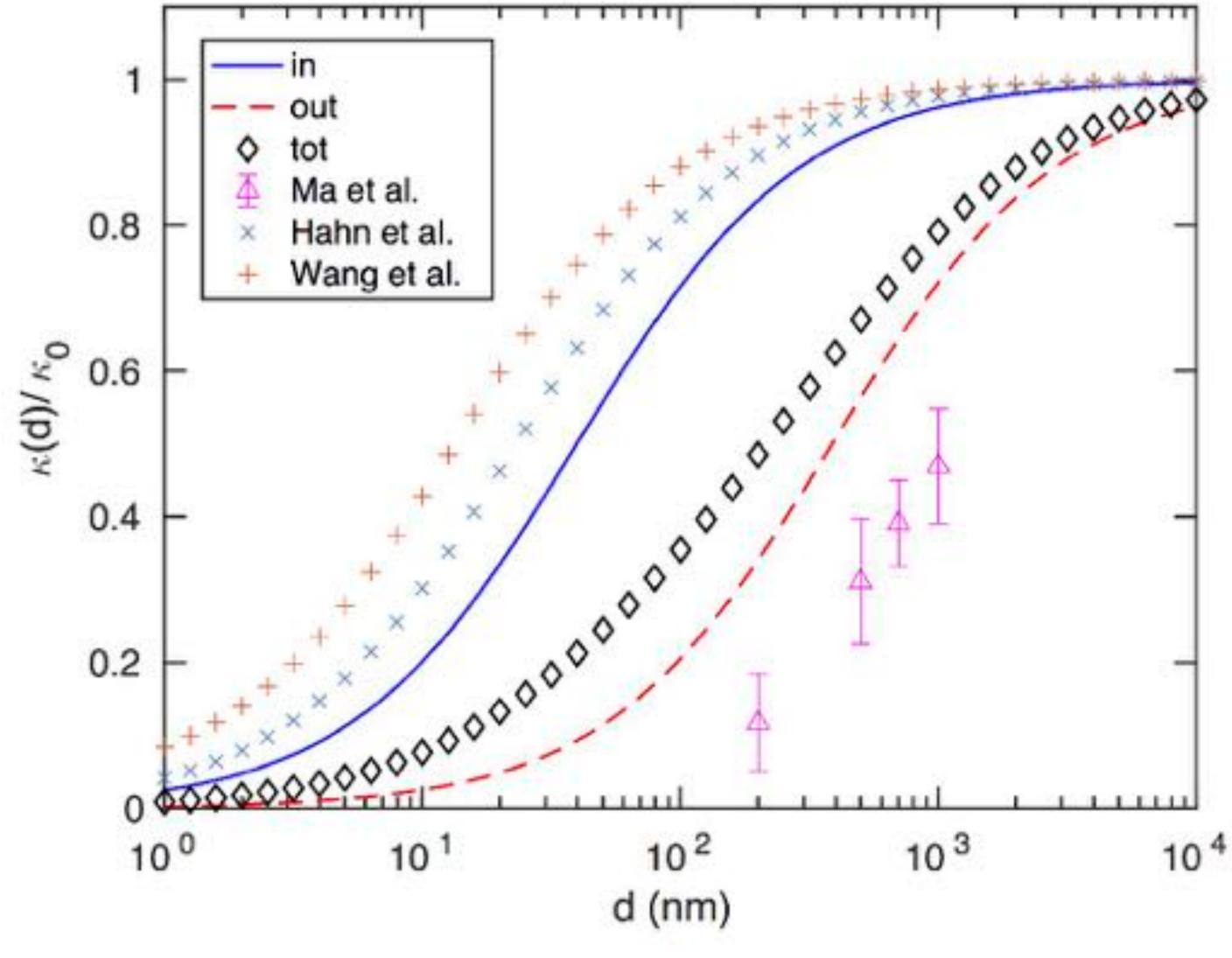






[Y. Wang *et al.*, J. Mater. Res. **29**, 362 (2014)]

[K.R. Hahn *et al.*, Carbon **96**, 429 (2016)]







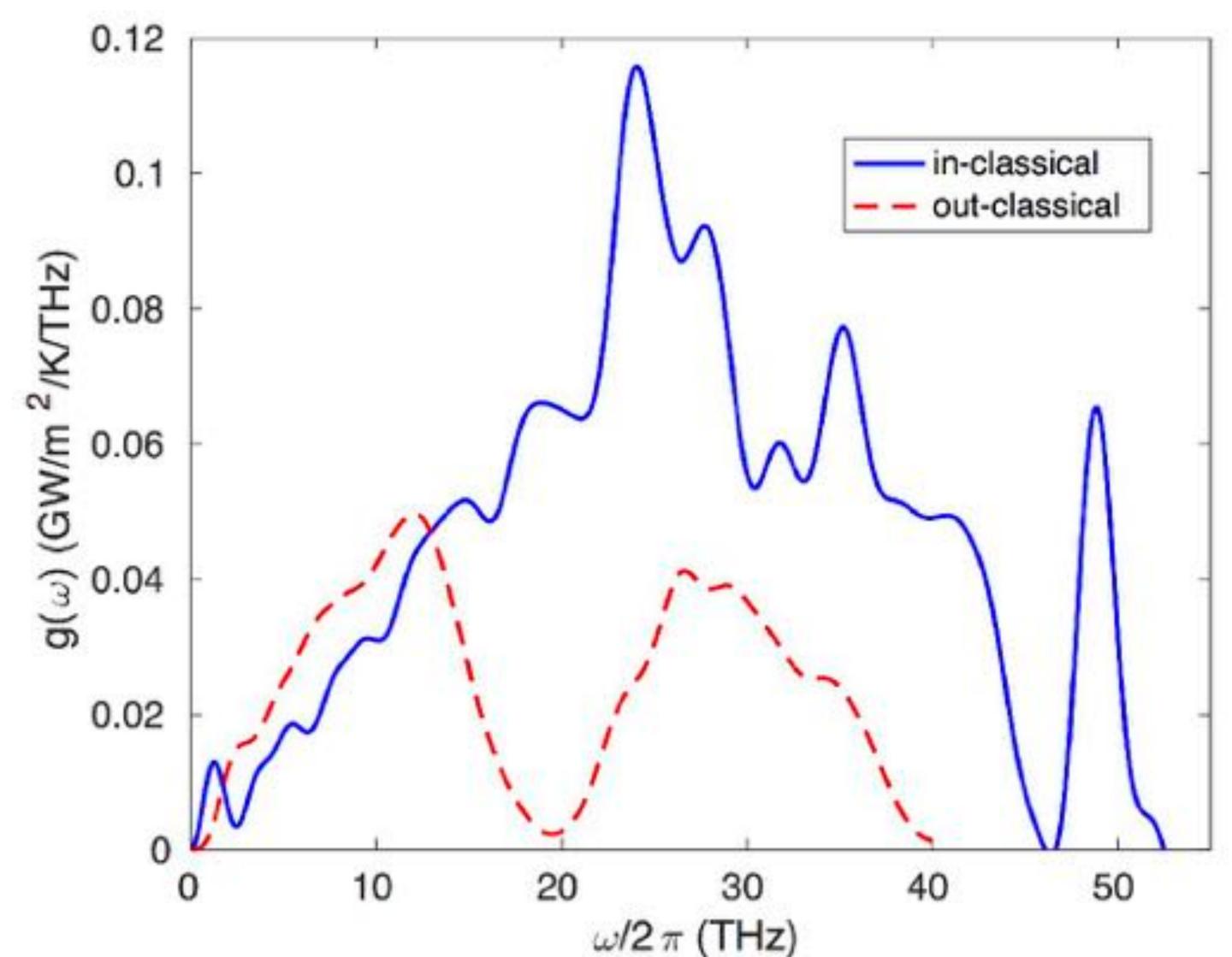




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Heat Conduction in Multigrain Flakes



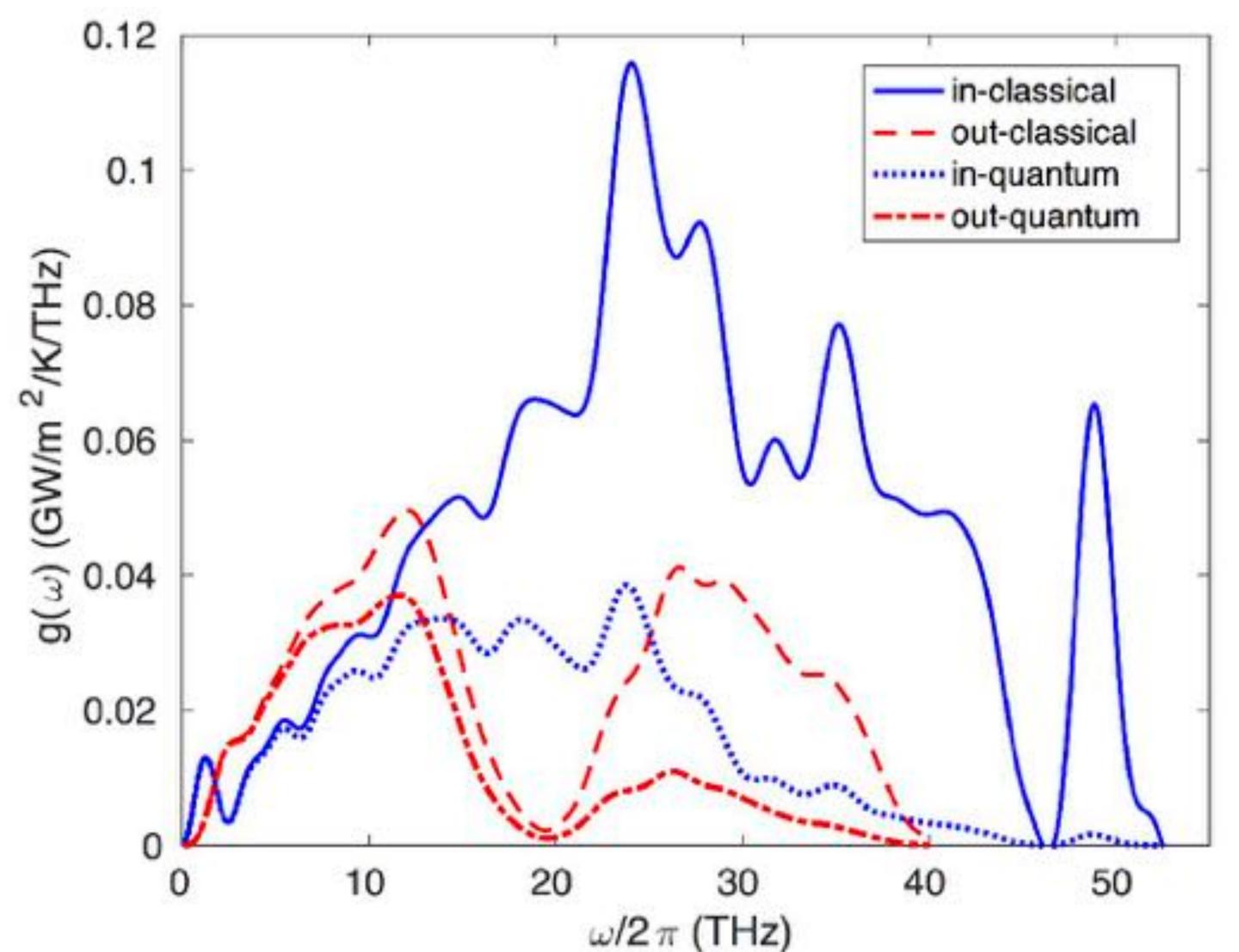
* Spectral conductance

$$g_{A \to B}^{\text{in/out}}(\omega) = \frac{q_{A \to B}^{\text{in/out}}(\omega)}{S|\Delta T|}$$

[K. Sääskilahti et al., AIP Adv. 6, 12190 (2016)]







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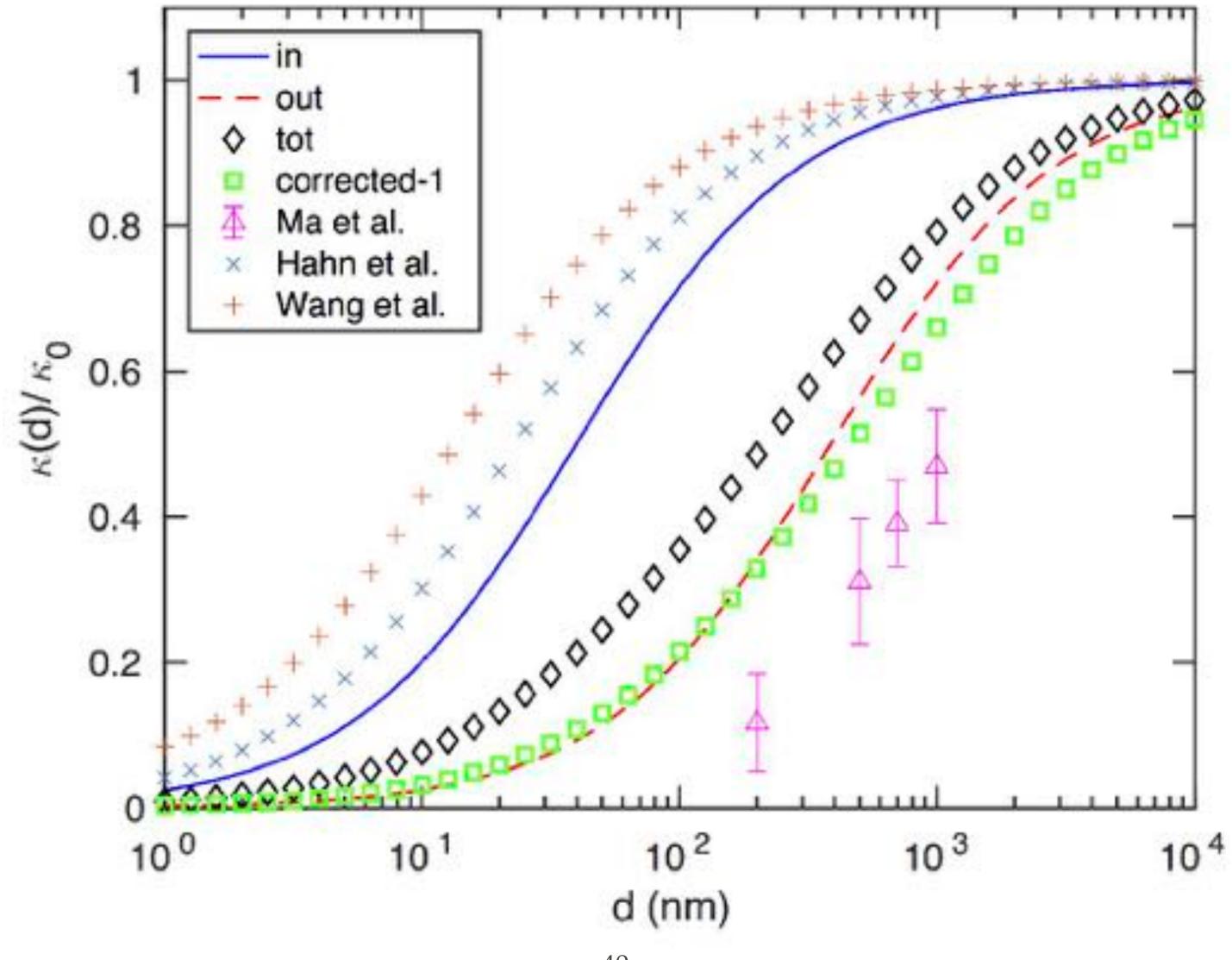
[K. Sääskilahti et al., AIP Adv. 6, 12190 (2016)]

* Quantum corrected by Bose-Einstein factor

$$L^{\rm in} \approx 120 \ {\rm nm} \ {\rm and} \ L^{\rm out} \approx 800 \ {\rm nm}$$











* For pristine graphene we find at T = 300 K (MD with opt. Tersoff)

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Experimentally $\kappa_0 \approx 1500 - 2500 \; \mathrm{Wm^{-1}K^{-1}}$





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$$\kappa_0 \approx 1500-2500~{\rm Wm^{-1}K^{-1}}$$
 $5~200~{\rm Wm^{-1}K^{-1}}$





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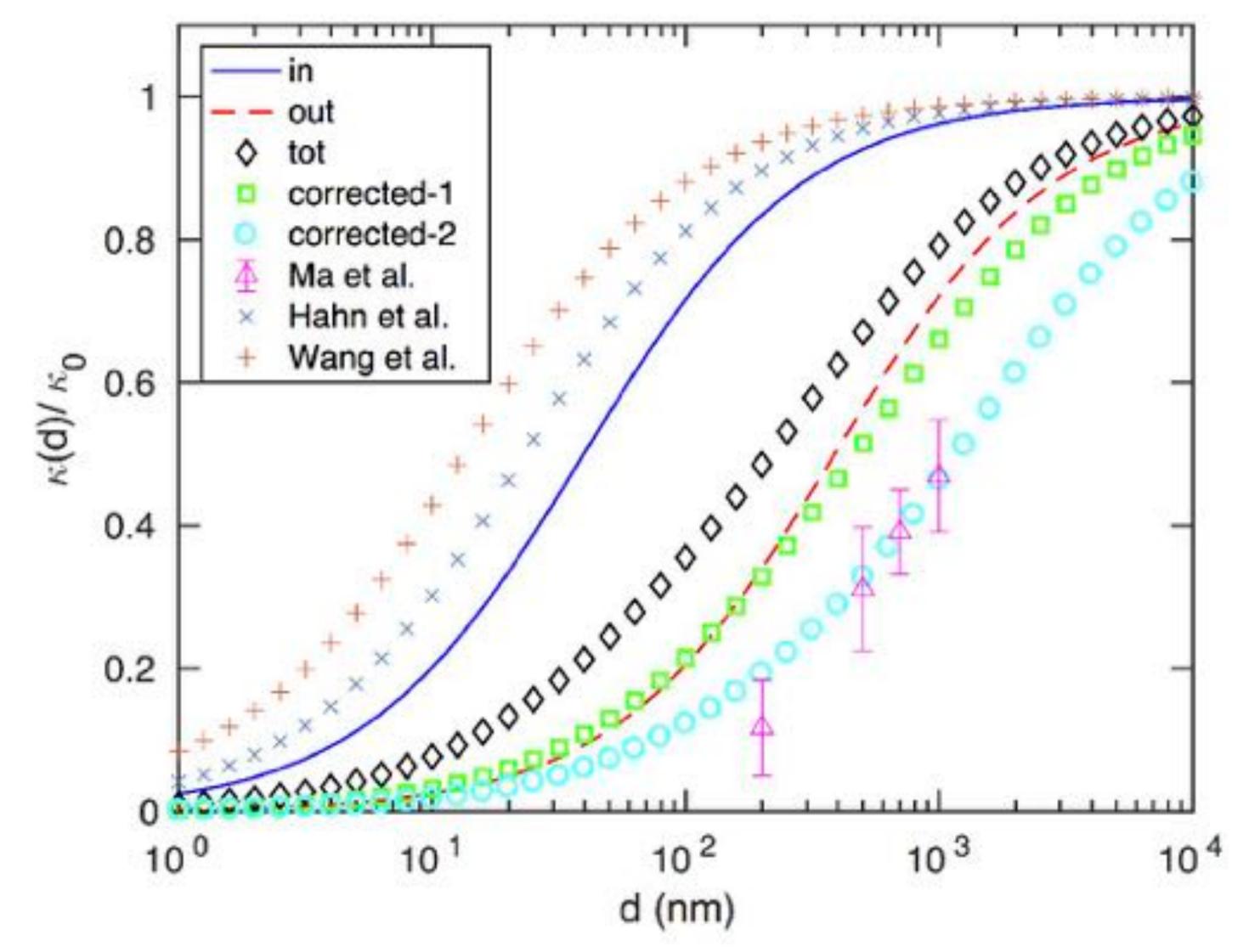
$$\kappa_0 = \kappa_0^{\text{in}} + \kappa_0^{\text{out}} = 2 900 \text{ Wm}^{-1} \text{K}^{-1}$$

Experimentally $\kappa_0 \approx 5 \ 200 \ \mathrm{Wm^{-1} K^{-1}}$

Recent lattice dynamics calculations give 5 450 W/mK [Y. Kuang et al., Int. J. Heat Mass Transfer 101, 772 (2016)]







* Final estimates for the Kapitza lengths

 $L^{\rm in} \approx 0.12~\mu{\rm m}$ and $L^{\rm out} \approx 2~\mu{\rm m}$





Summary and Conclusions

- * Phase-field crystal models can be employed for a *quantitative description* of 2D grain boundaries in graphene by proper fitting of the elastic properties
- * PFC models produce realistic 5 | 7 grain boundaries in most cases
- * Large multigrain samples (microns in linear size) can be generated for MD relaxation in 3D and used for further investigations (thermal, mechanical and transport properties) thermal transport is controlled by flexural modes in pristine graphene and strongly affected by grains
- * Quantum corrections need to be taken into account both for pristine and multigrain graphene (because of high Debye temperature)





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